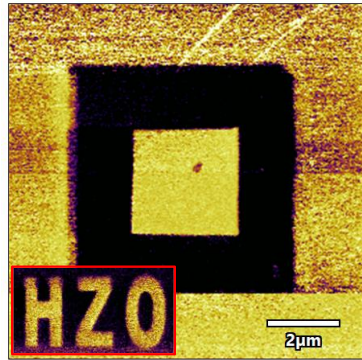
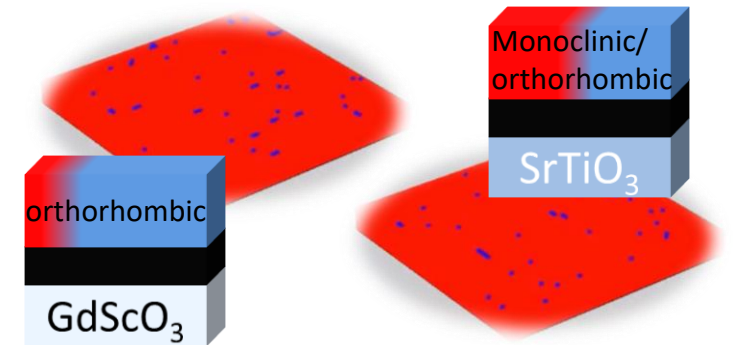
 "la Caixa" Foundation



Worse might be better in ferroelectric HfO_2

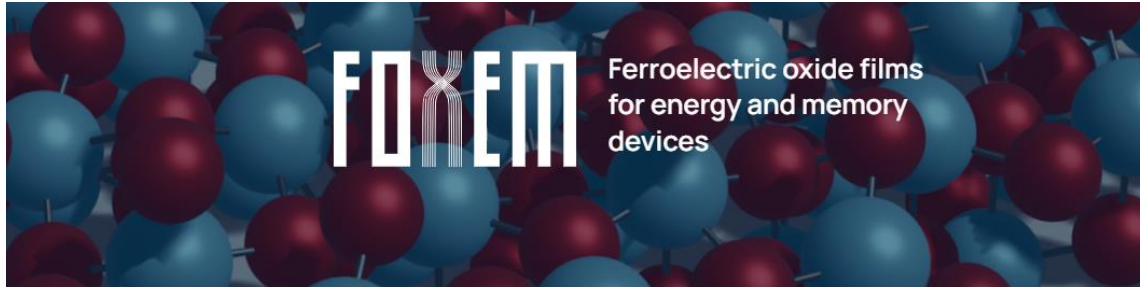


T. Song, H. Tan,
F. Sánchez, Ignasi Fina

ignasifinamartinez@gmail.com

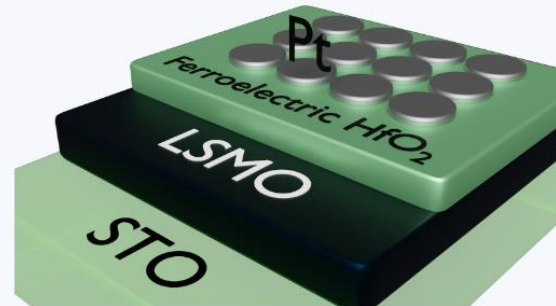
ifina@icmab.es

Institut de Ciència de Materials de Barcelona (ICMAB-CSIC), Campus UAB, 08193 Barcelona, Spain

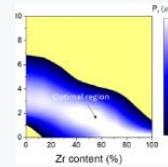


SCOPE

Ferroelectric oxide films for energy and memory devices group (FOXEM) aims to develop high quality new ferroelectric materials compatible with industry to study their properties from a fundamental to a device level. Electronics Industry is facing several bottlenecks to sustain the increasing demand and necessity of new data storage, computation and communication devices. New materials are needed and CMOS-compatible ferroelectrics based in HfO₂ are in the spotlight. We investigate epitaxial oxide thin films of these oxides as model systems to understand and improve the ferroelectric properties. Our activities involve growth, structural studies, advanced characterization of electrical properties and prototyping of conventional and emerging memory devices.

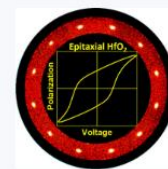


SELECTED WORKS



Ferroelectric (Hf,Zr,La)O₂ films

T. Song, S. Estandia, I. Fina, and F. Sánchez, Ferroelectric (Hf,Zr,La)O₂ films. Applied Materials Today 29 (2022). <https://doi.org/10.1016/j.apmt.2022.101661>



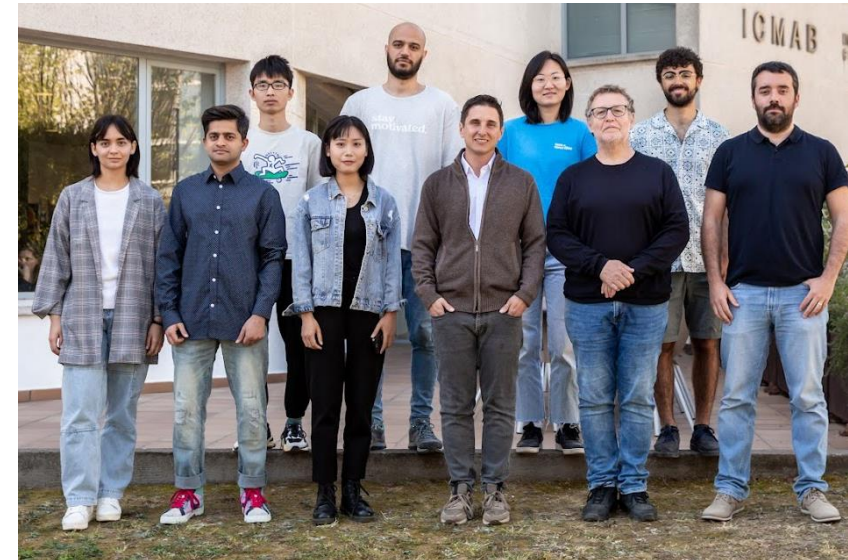
Epitaxial Ferroelectric HfO₂ Films: Growth, Properties, and Devices

I. Fina and F. Sánchez, Epitaxial Ferroelectric HfO₂ Films: Growth, Properties, and Devices. ACS Appl. Electron. Mater. 3, 1530 (2021). <https://doi.org/10.1021/acsaem.1c00110>



Control of up-to-down/down-to-up light-induced ferroelectric polarization reversal

H. Tan, G. Castro, J. Lyu, P. Loza-Alvarez, F. Sánchez, J. Fontcuberta, and I. Fina, Control of up-to-down/down-to-up light-induced ferroelectric polarization reversal. Materials Horizons 9, 2345 (2022). <https://doi.org/10.1039/d2mh00644h>

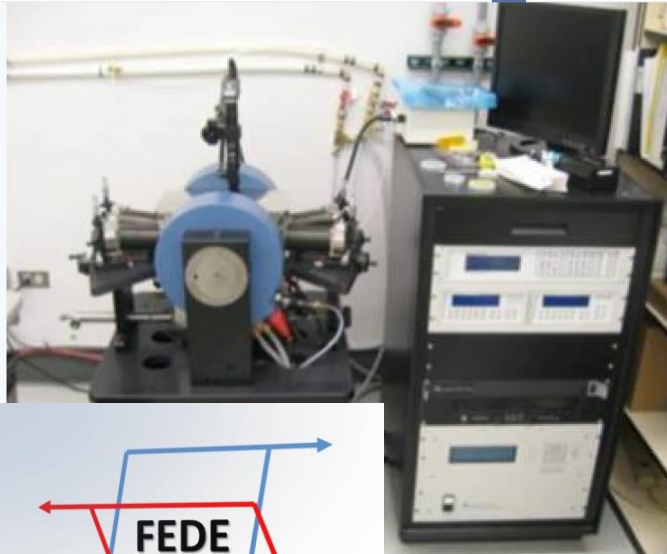


<https://foxem.icmab.es/>

Ferroelectric and dielectric laboratory

LABORATORY OF
MULTIFUNCTIONAL THIN FILMS
AND COMPLEX STRUCTURES
INSTITUT DE CIÈNCIES DE MATERIALS DE BARCELONA, CSIC-ICMAB

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performed:

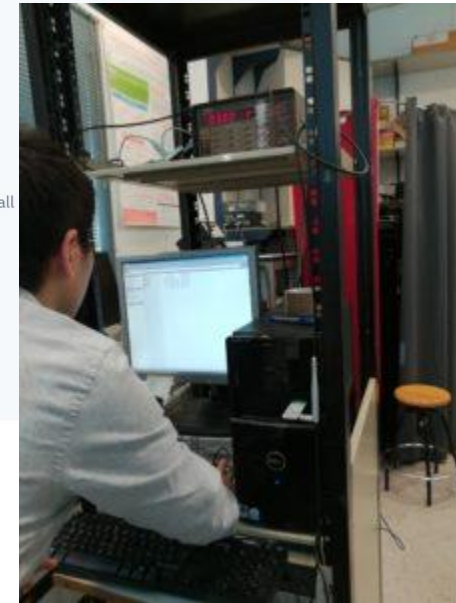
- Impedance spectroscopy $Z(\omega)$, for $\omega < 1\text{MHz}$, look in I. Fina, et al., Thin Solid Films 518, 4710(2010).

LABS

FERroelectric and DiElectric Laboratory

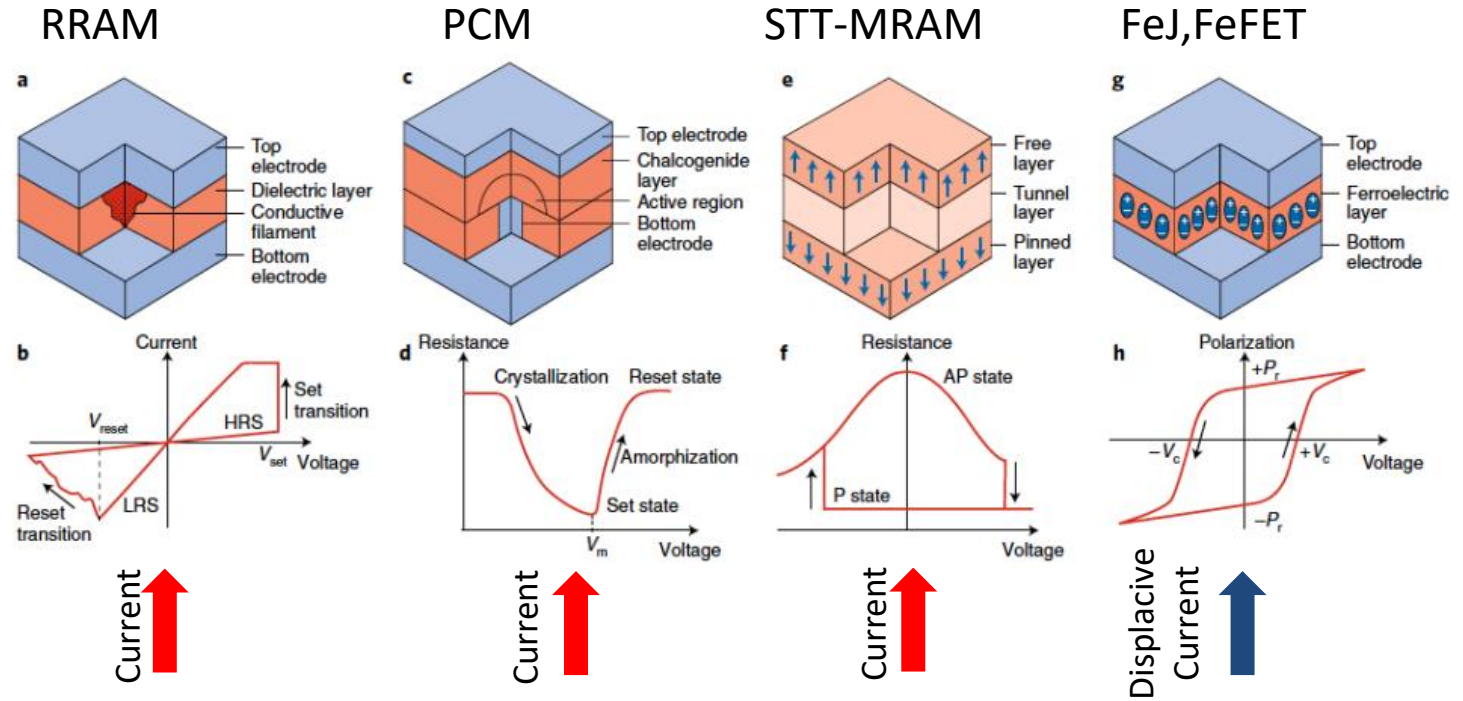
Our FERroelectric/DiElectric laboratory is specially designed to characterize all of functional properties of ferroelectric and multiferroic thin films

Laboratory lead and designed by Dr. Ignasi Fina



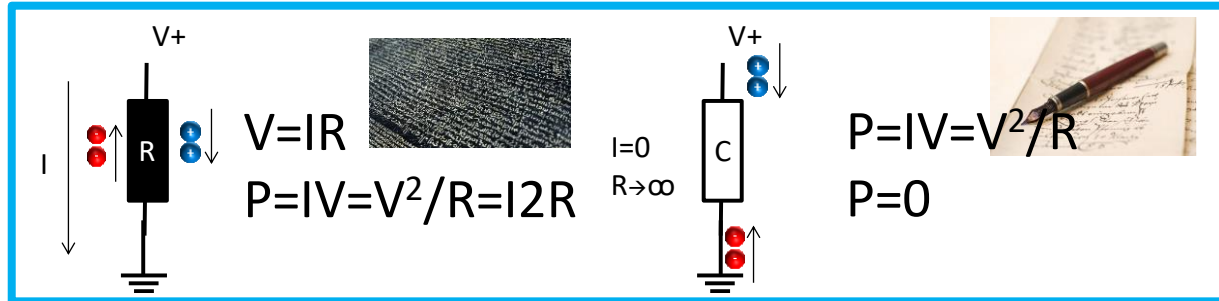
Introduction: Present and future for memristors

Current status



J. J. Yang *et al*, *Nat. Nanotechnol.* 2008.
 R. Waser *et al*, *Adv. Mater.* 2009.
 D. Ielmini *et al*, *Phase Transitions* 2011.
 D. Ielmini *et al*, *Nature Electronics*
[AnChen](#)
 IBM Research, San Jose, CA 95120, USA

Future? →

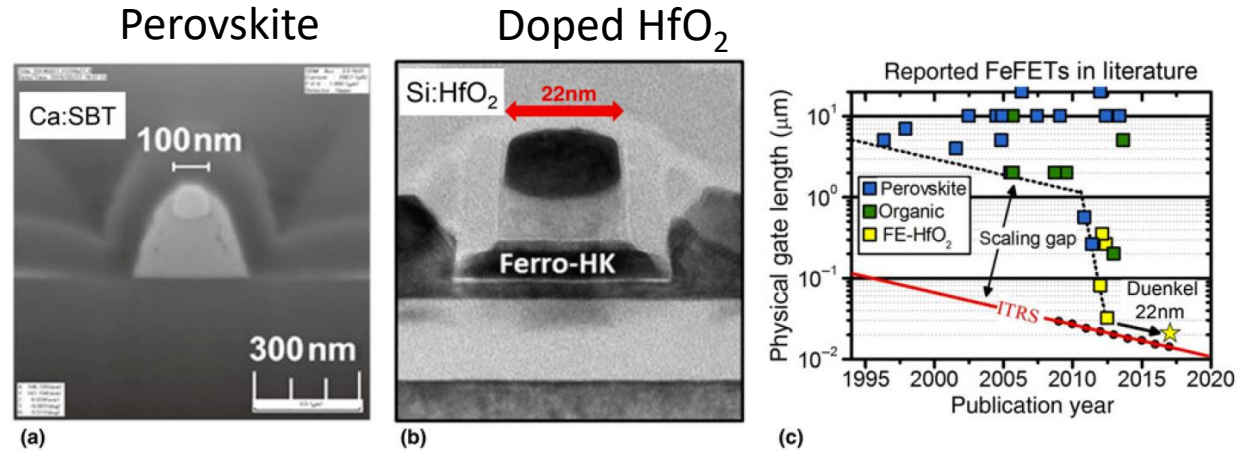
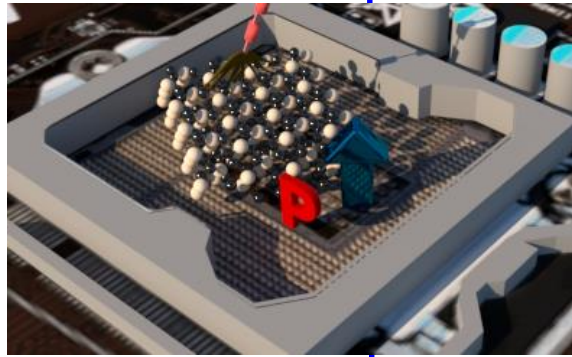


Ferroelectric means pure electronic effects

- Efficient
- Fast
- Reliable

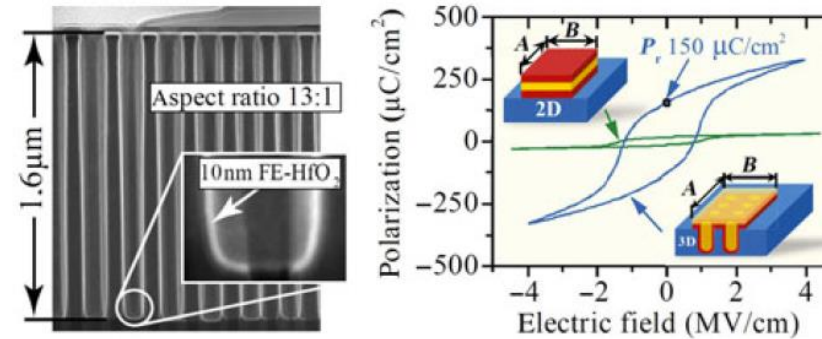
Introduction: Ferroelectric hafnium oxide

Ferroelectric FETs



REVIEW: Park et al., MRS Comm. 8, 795 (2018)

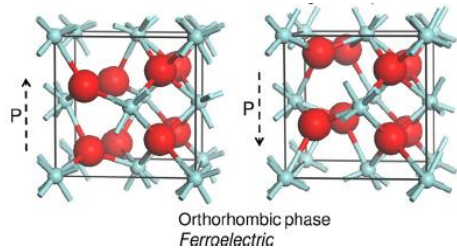
3D capacitors, $t = 10$ nm



REVIEW: Park et al., MRS Comm. 8, 795 (2018)

Introduction: ferroelectricity in hafnium oxide

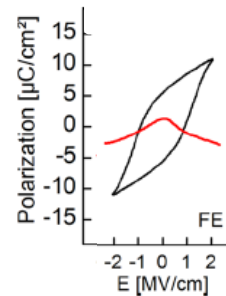
Polycrystalline doped HfO_2



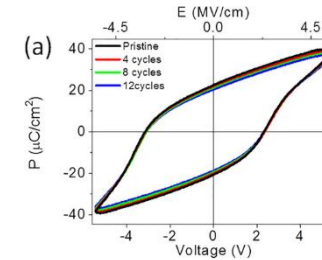
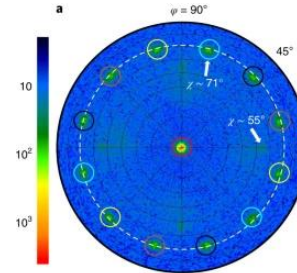
Orthorhombic phase

is

Ferroelectric

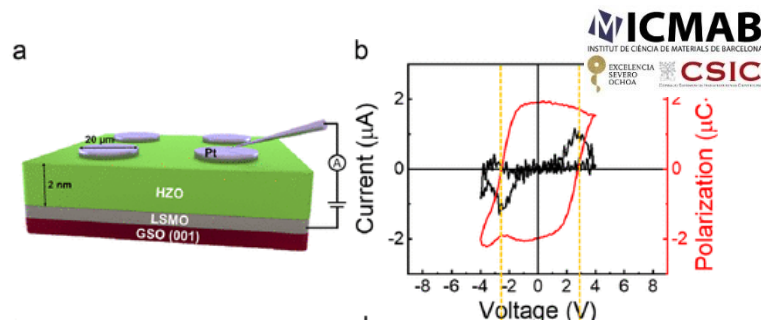


Epitaxial $\text{Hf}_{0.5}\text{Zr}_{0.5}\text{O}_2$

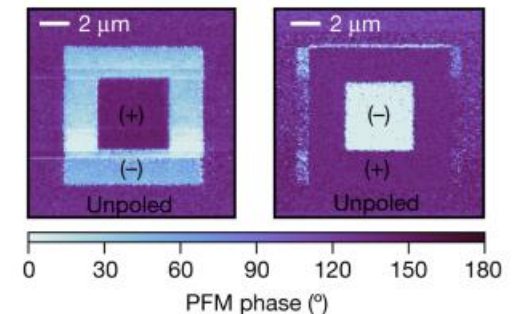
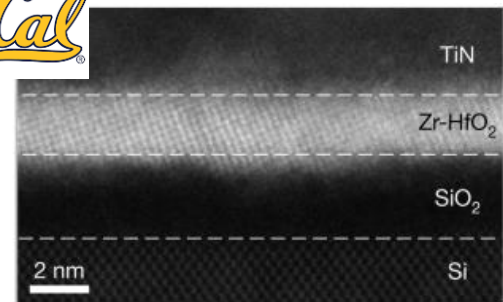


Y. Wei, ... B. Noheda. *Nature Materials* **17**, 1095–1100 (2018).
 J.Lyu. *IF*, ..., F. Sánchez, *Appl. Phys. Lett.* **113**, 082902 (2018);
ACS Applied Electronic Materials **1**, 220 (2019); *Appl. Phys. Lett.* In-print (2019).
 Katayama, ..., Funakubo *APL* **109**, 112901 (2016); Shimizu et al., *Sci. Rep.* **6**, 32931 (2016)

Ferroelectric response in epitaxial ultrathin films



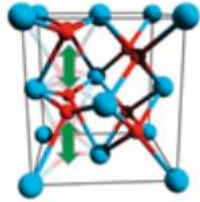
Cal



S. S. Cheema, et al., *Nature* **580**, 478 (2020); *Nature* **604**, 7904 (2022)
 M.C. Sulzbach, *IF*, et al., *ACS Applied Electronic Materials* **3** (8), 3657–3666 (2021)

Introduction: Ferroelectric hafnium oxide

Orthorhombic HfO_2



Ferroelectric hafnium oxide for ferroelectric random-access memories and ferroelectric field-effect transistors

Thomas Mikolajick, Stefan Slesazeck, Min Hyuk Park, and Uwe Schroeder

the adoption of ferroelectric hafnium oxide.

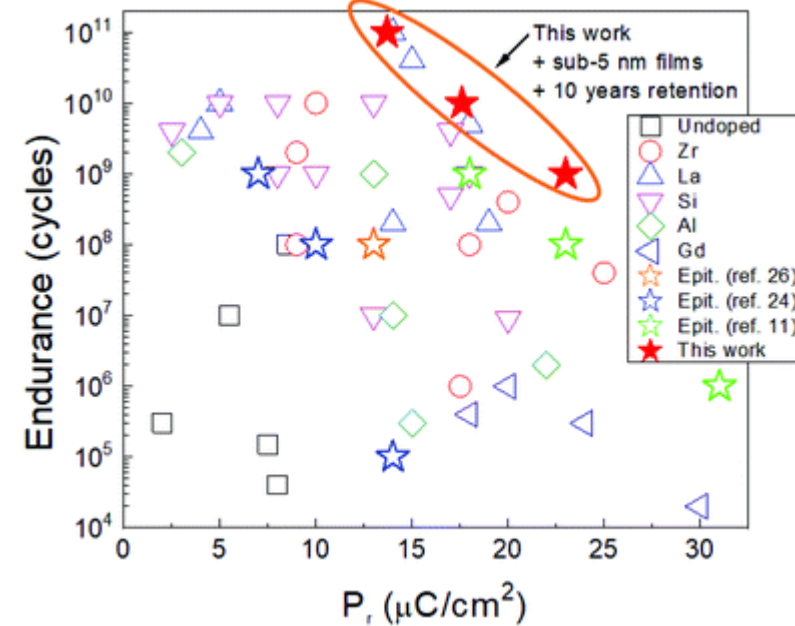
Nevertheless, achieving a cycling endurance beyond the level of conventional charge-based nonvolatile memories remains a challenge.^{39,63} Different strategies have been proposed to overcome these limitations⁶⁴ and encouraging results have recently

Table I. Comparison of coercive field, E_c , and switched polarization charge, $2P_r$, for strontium bismuth tantalate (SBT), lead zirconium titanate (PZT), poly(vinylidene fluoride):tetrafluoroethylene (PVDF-TRFE), and doped hafnium oxide.

	SBT ($\text{Sr}_2\text{Bi}_2\text{TaO}_9$)	PZT	PVDF-TRFE	Doped HfO_2
Coercive field E_c in MV/cm	0.05	0.1	0.5	0.8–2
Switched charge ($2P_r$) in $\mu\text{C}/\text{cm}^2$	15–25	30–60	10	30–60

T. Mikolajick, et al. MRS Bull. 43, 340 (2018)

Polarization-endurance dilemma

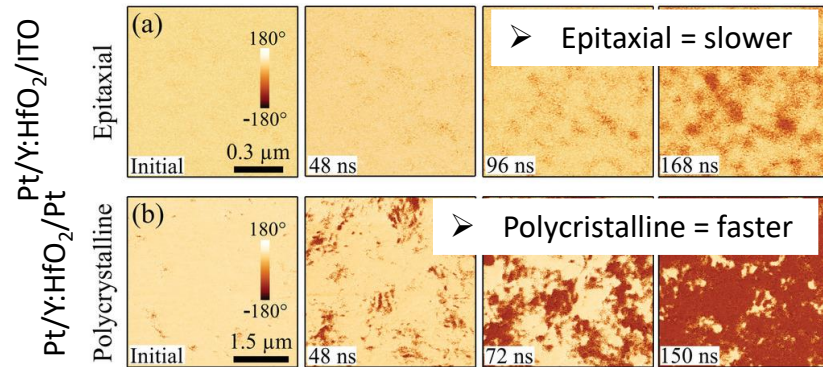


- Best endurance is NOT shown by films showing larger P_r

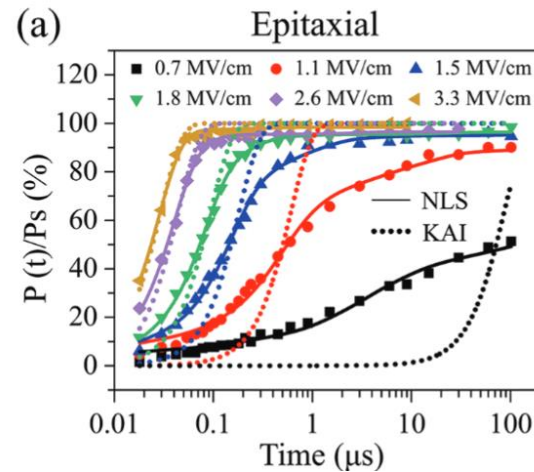
J. Lyu, ..., IF, Sánchez, Nanoscale 12, 11280 (2020)

Introduction: switching dynamics in polycrystalline film

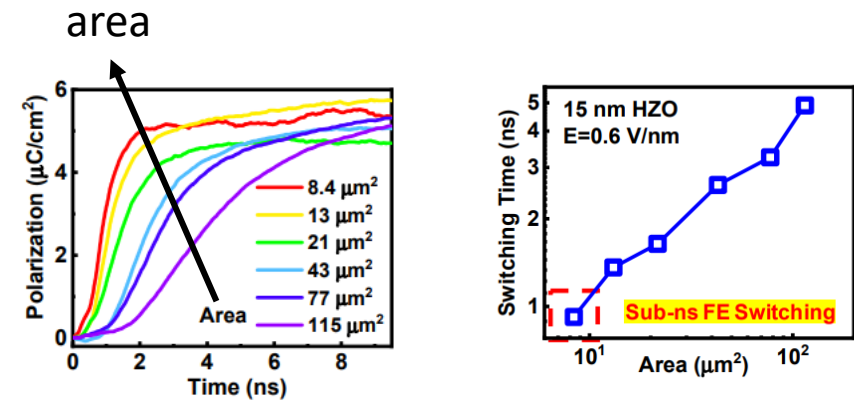
Nucleation-limited switching model (NLS)



- As expected for a polycrystalline film
- KAI model is not valid.



Subnanosecond switching speed (at least)



- 925 ps
- Important contribution of RC time
- Phase coexistence is present

P. Buragohain, et al, *Adv. Funct. Mater.*, 2022, **32**, 2108876.

X. Lyu et al., presented at the 2019 IEEE Int. Electron Devices Meeting (IEDM), 2019.

A. K. Tagantsev, et al, *Phys. Rev. B-Condens. Matter Mater. Phys.*, 2002, **66**, 1-6.

J. Y. Jo, et al, *Phys. Rev. Lett.*, 2007, 99, 1-4.

Introduction: Epitaxial ferroelectric hafnium oxide

Material	deposition			Substrate	Top/bottom electrode	Thickness (nm)	P_r ($\mu\text{C}/\text{cm}^2$)	E_c (MV/cm)	Endurance (cycles)/ E_{cycling} (MV/cm)	Retention/ E_{poiling} (MV/cm)	ref
	Method	Temperature ($^{\circ}\text{C}$)	Atmosphere								
Hf _{0.93} Y _{0.07} O ₂	PLD	700 °C	0.01 Torr (O ₂)	YSZ(110)	Pt/ITO	15	~12	~2			Shimizu ³⁷
Hf _{0.93} Y _{0.07} O ₂	PLD	700 °C	0.01 Torr (O ₂)	YSZ(111)	Pt/ITO	14	~10	~2			Katayama ³⁸
Hf _{0.93} Y _{0.07} O ₂	Sputtering	RT + 1000 °C annealing	0.2 Torr (Ar)	YSZ(111)	Pt/ITO	24	~11	~2.2			Suzuki ⁴⁵
Hf _{0.93} Y _{0.07} O ₂	PLD	RT + 1000 °C annealing	0.01 Torr (O ₂)	YSZ(111)	Pt/ITO	15	15	~2.1			Mimura ⁴⁴
Hf _{0.93} Y _{0.07} O ₂	PLD	RT + 1000 °C annealing	0.01 Torr (O ₂)	YSZ(111)	*/ITO	111	~5	~1.4			Mimura ⁸⁰
Hf _{0.93} Y _{0.07} O ₂	Sputtering	RT + 800 °C annealing	0.2 Torr (Ar)	YSZ(111) and -(001)	*/ITO	380 and 1080	~5	~1			Shimura ⁹¹
Hf _{0.94} Fe _{0.06} O ₂	Ion beam	RT + 900 °C annealing	3.8 × 10 ⁻⁵ Torr	YSZ(001)	Pt/ITO	20	8.8	~2			Shiraishi ⁴¹
Hf _{0.9} Ce _{0.1} O ₂	Ion beam	RT + 900 °C annealing	3.8 × 10 ⁻⁵ Torr	YSZ(001)	Pt/ITO	30	~5				Shiraishi ⁴²
Hf _{0.93} Y _{0.07} O ₂	Sputtering	RT	0.01 Torr (Ar)	YSZ(111)	Pt/ITO	16	15	2.3			Mimura ⁴⁸
Hf _{1-x} Y _x O ₂ (x = *)	PLD	700 °C	0.15 Torr (O ₂)	YSZ/Si(001)	Pt/-	*	~20 (leaky)	*			Lee ⁹⁰
Hf _{0.936} Si _{0.044} O ₂	PLD	700 °C	0.1 Torr (O ₂)	Nb:STO(111) and -(110)	Au-Cr/substrate	3-15	up to ~32	4-5			Li ⁵⁰
Hf _{0.5} Zr _{0.5} O ₂	PLD	700 °C		YSZ(111) and -(110)	Au-Cr/TiN	15	~7-20	1.1-2.3			Li ⁴⁰
Hf _{0.5} Zr _{0.5} O ₂	PLD	800 °C	0.1 mbar	STO(001)	Pt/LSMO	9	20	3	1 × 10 ⁸ (5)	>10 (6.1)	Lyu ⁵²
Hf _{0.5} Zr _{0.5} O ₂	PLD	800 °C	0.1 mbar	STO(001)	LSMO/LSMO	5	34	~5			Wei ⁵¹
Hf _{0.5} Zr _{0.5} O ₂	PLD	800 °C	0.1 mbar	STO(001)	LSMO/LSMO	9	18	~3	1 × 10 ⁵ (4.4)		Wei ⁵¹
Hf _{0.5} Zr _{0.5} O ₂	PLD	550 °C	0.13 mbar	LAO(001)	Pd/LSMO	10	20	2.4			Yoong ⁷⁸
Hf _{0.5} Zr _{0.5} O ₂	PLD	800 °C	0.1 mbar	YSZ/Si(001)	Pt/LSMO	4.6	33	~4	1 × 10 ¹¹ (5.4)	>10 (5.4)	Lyu ¹¹⁶
Hf _{0.5} Zr _{0.5} O ₂	PLD	800 °C	0.1 mbar	STO/Si(001)	Pt/LSMO	7.7	34	~3	1 × 10 ⁹ (5.2)	>10 (5.2)	Lyu ⁶⁶
Hf _{0.5} Zr _{0.5} O ₂	PLD	800 °C	0.1 mbar	GdScO ₃ and TbScO ₃ (001)	Pt/LSMO	9	~24	~2.5			Estandia ⁶⁴
Hf _{0.5} Zr _{0.49} La _{0.01} O ₂	PLD	800 °C	0.1 mbar	STO(001)	Pt/LSMO	4.8	~20	~3.7	5 × 10 ¹⁰ (5.4)	>10 (5.4)	Song ⁶³
Hf _{0.5} Zr _{0.49} La _{0.01} O ₂	PLD	800 °C	0.1 mbar	STO/Si(001)	Pt/LSMO	6.3	~30	~3.5	1 × 10 ⁹ (4.3)	>10 (7.2)	Song ⁶³
Hf _{0.945} La _{0.055} O ₂	PLD	600 °C	0.1 mbar	STO(001)	Pt/LSMO	12	~16	~2.7	2 × 10 ⁷ (5.3)	>10 (5.3)	Li ⁸⁸

- ICMAB systematic studies on endurance and retention are unique.
- Other reported results are far from the best performance of ICMAB films

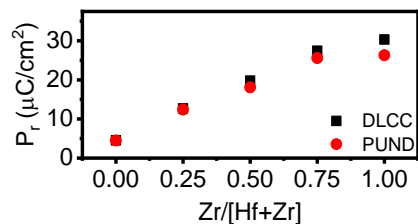
Review: Fina and Sánchez, ACS Applied Electronic Materials 3, 1530 (2021)

Introduction: Epitaxial ferroelectric hafnium oxide, taxonomy

Strategies to improve functional properties in epitaxial films

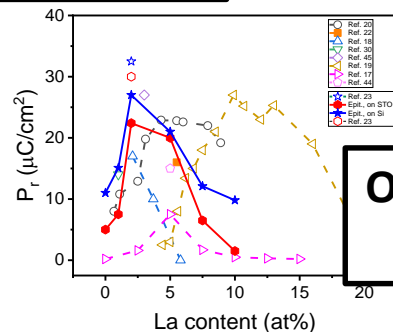
1 Doping

Zr doping



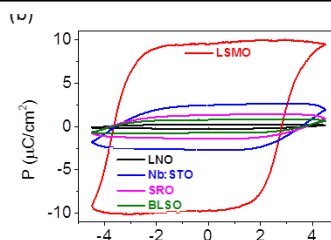
Orthorhombic
Tetragonal

La doping



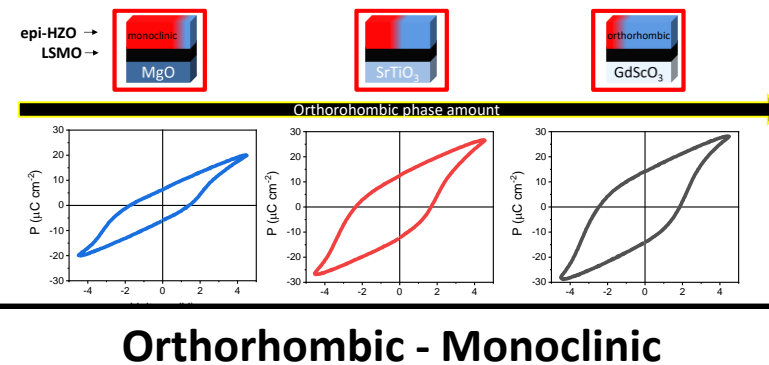
Orthorhombic
Cubic

2 Electrode selection



Orthorhombic
Monoclinic

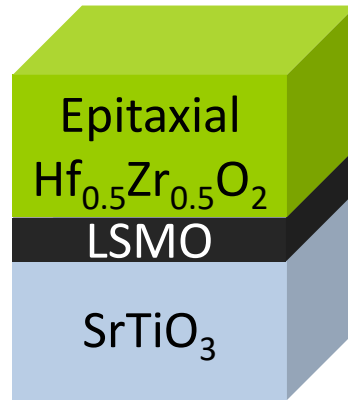
3 Epitaxial stress



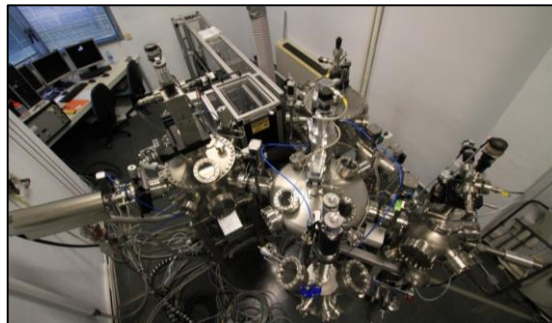
Orthorhombic - Monoclinic

Kiguchi et al., J. Ceram. Soc. Jpn. 124, 689 (2016)
Mimura et al., Jpn. J. Appl. Phys. 59, SGGB04 (2020)
Shimizu et al., Appl. Phys. Lett. 2015, 107 (3), 032910
Müller et al., Nano Lett. 2012, 12, 4318–4323
Kozodaev, et al., J. Appl. Phys. 125, 034101 (2019)

Introduction: ferroelectricity in epitaxial HZO (homemade)



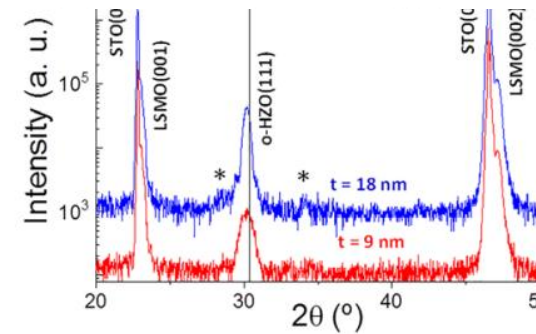
Epitaxial heterostructures
 fabricated by PLD in a single
 process



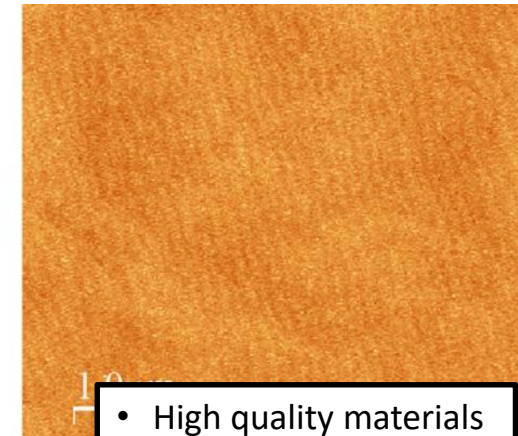
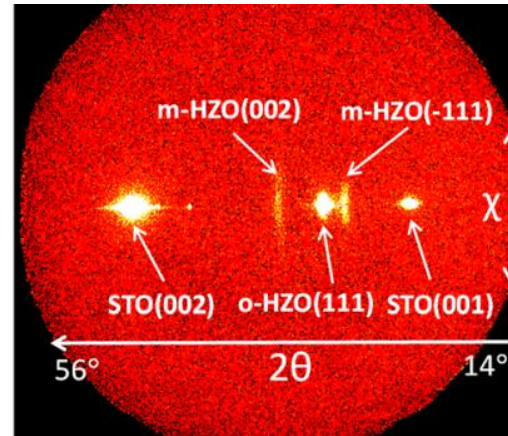
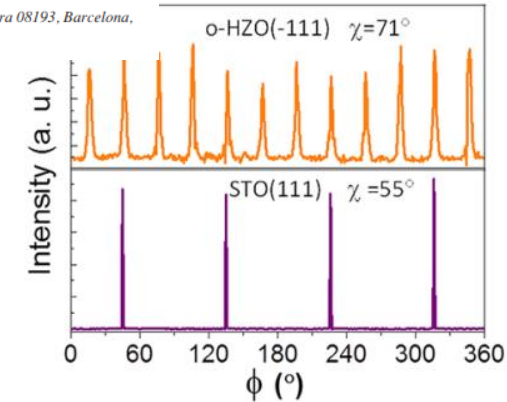
PLD Lab at ICMAB

Robust ferroelectricity in epitaxial $\text{Hf}_{1/2}\text{Zr}_{1/2}\text{O}_2$ thin films

J. Lyu, I. Fina, R. Solanas, J. Fontcuberta, and F. Sánchez^{a)}
 Institut de Ciència de Materials de Barcelona (ICMAB-CSIC), Campus UAB, Bellaterra 08193, Barcelona,
 Spain



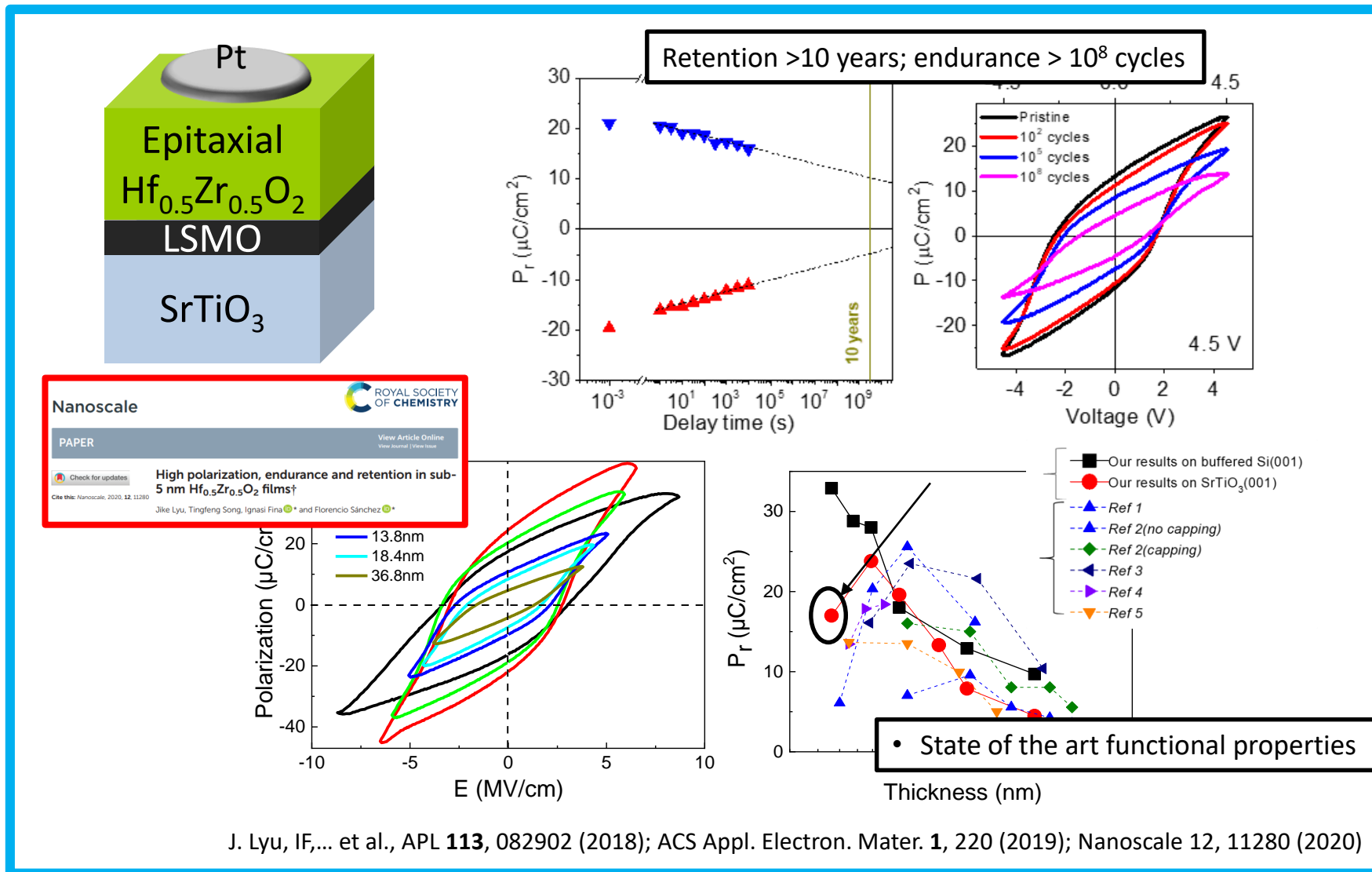
thickness: ~9 nm



- High quality materials
- Sharp interfaces

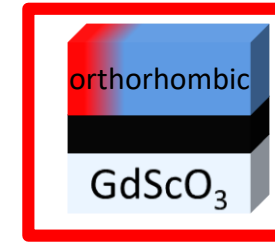
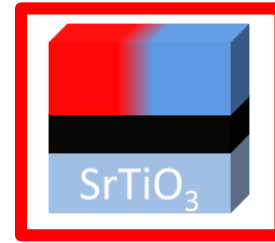
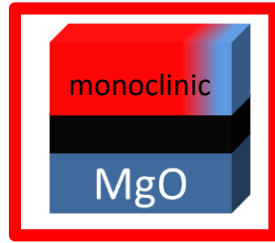
Lyu, IF, F. Sánchez
 Appl. Phys. Lett. 113, 082902 (2018)

Introduction: ferroelectricity in epitaxial HZO (homemade)



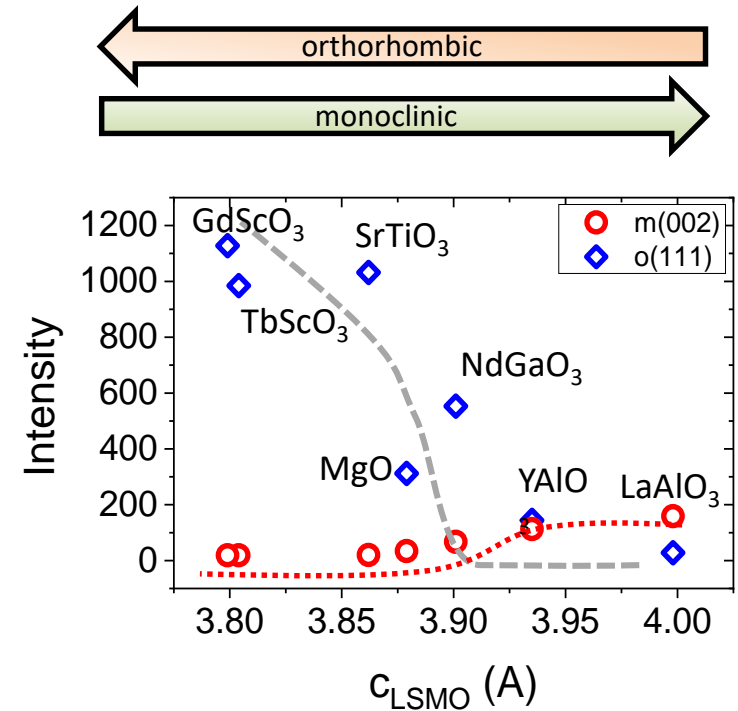
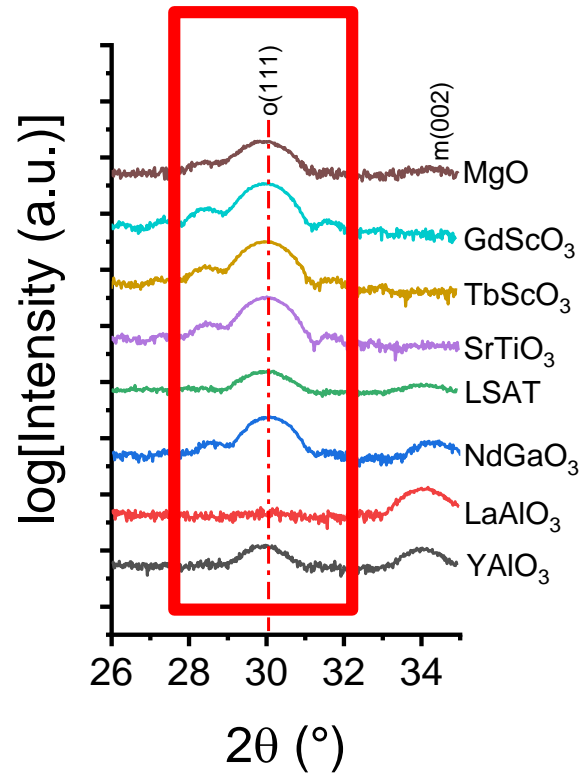
Results: tuning of orthorhombic phase in epitaxial LHO (PLD films)

epi-La:HfO₂ →
LSMO →



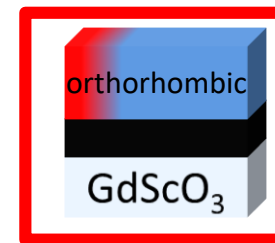
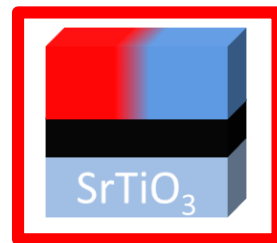
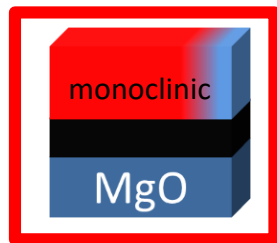
- All films are relaxed
- There are not variations of lattice parameters depending on the substrate
- You can't change the phase by strain

- The orthorhombic phase ratio depends on the selected substrate

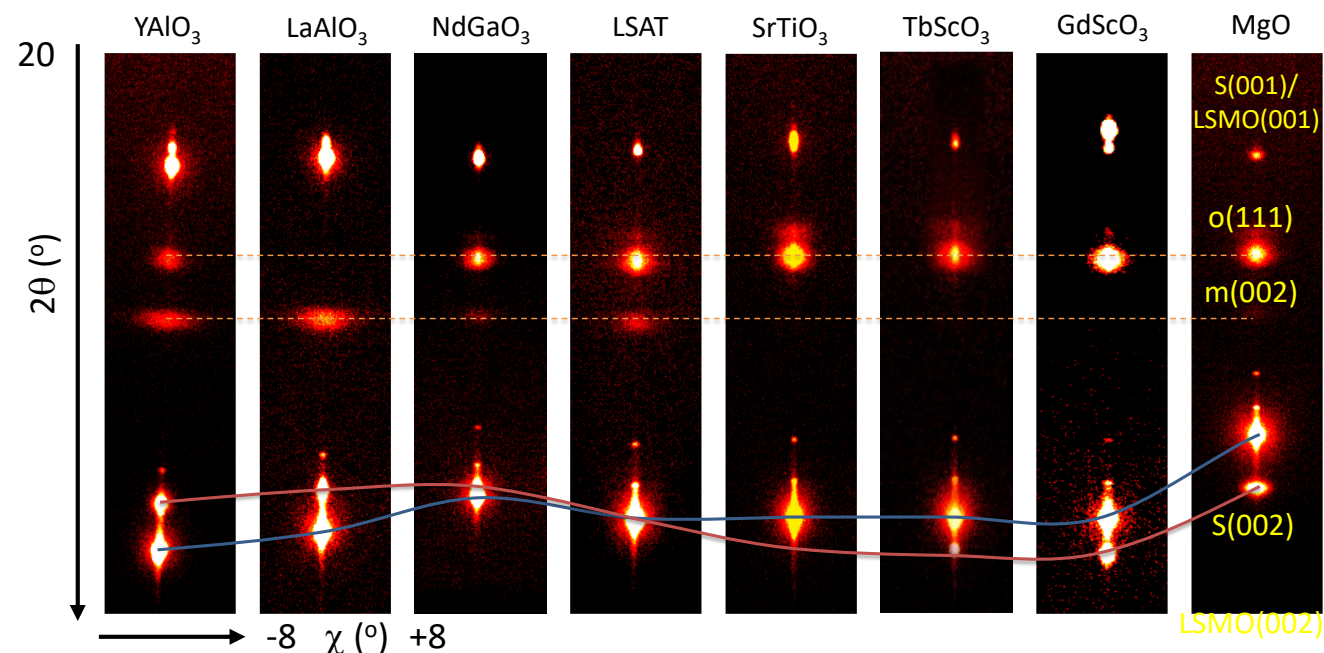


Results: tuning of orthorhombic phase in epitaxial LHO (PLD films)

epi-La:HfO₂ →
LSMO →



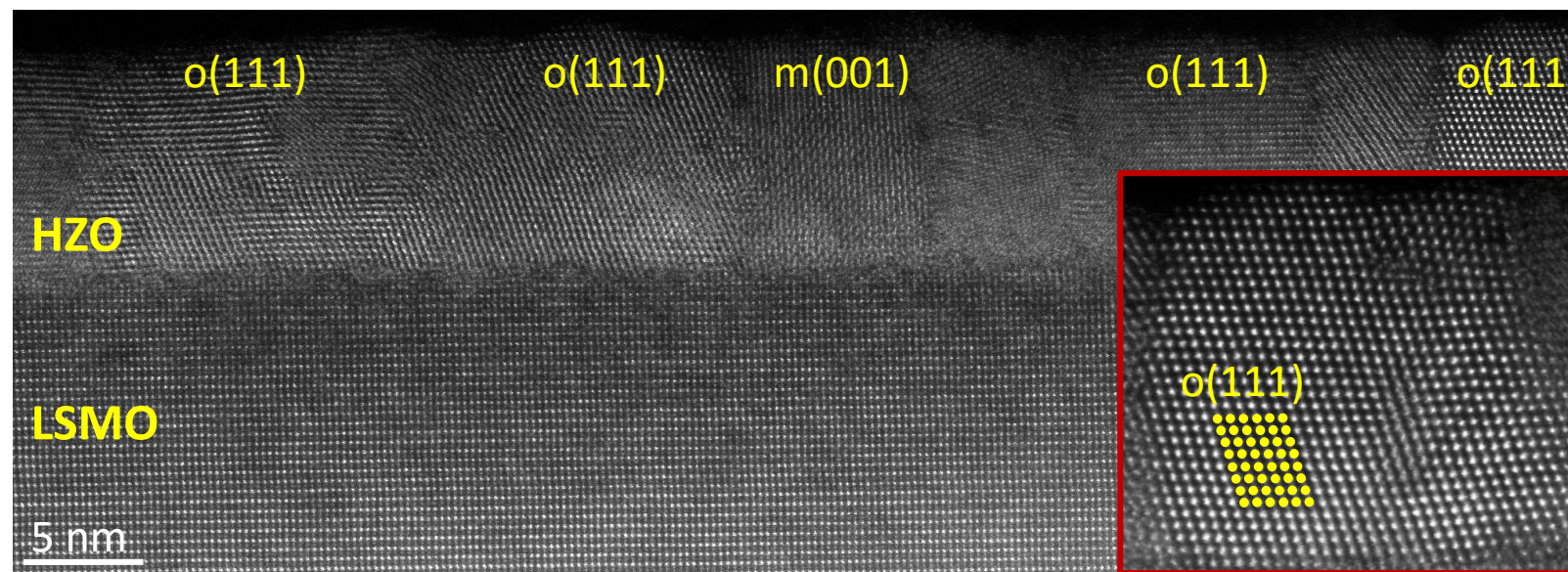
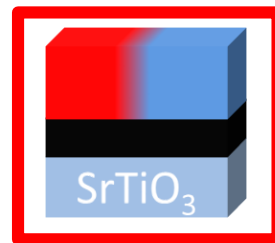
Orthorhombic phase amount



Substrate in-plane pseudo-cubic or cubic lattice parameter

Results: tuning of orthorhombic phase in epitaxial LHO (PLD films)

epi-La:HfO₂ →
LSMO →

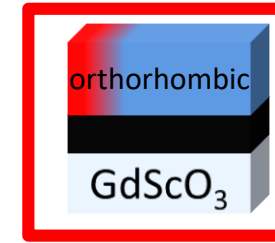
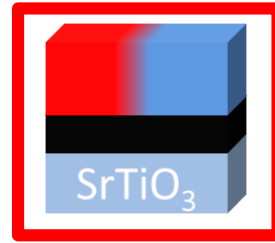
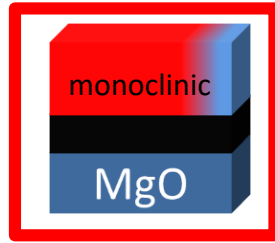


Coexistence of monoclinic (001)
and orthorhombic (111) grains

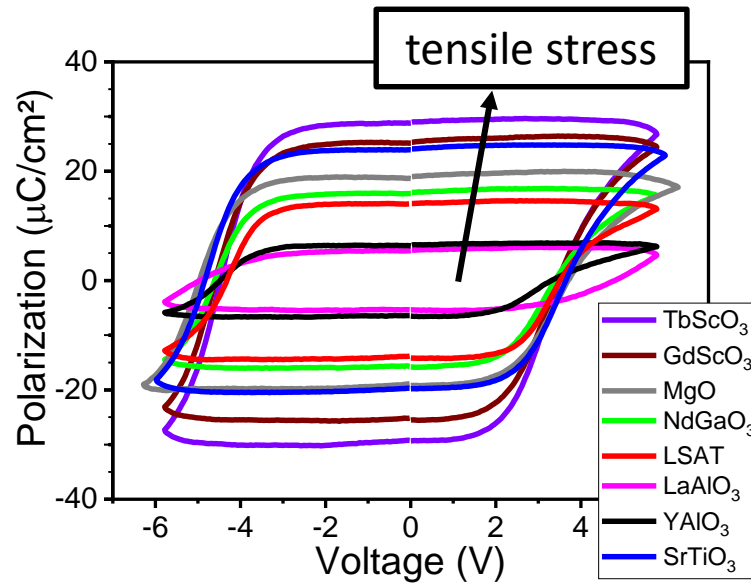
Collaboration S Estandia, J Gazquez

Results: tuning of orthorhombic phase in epitaxial LHO (PLD films)

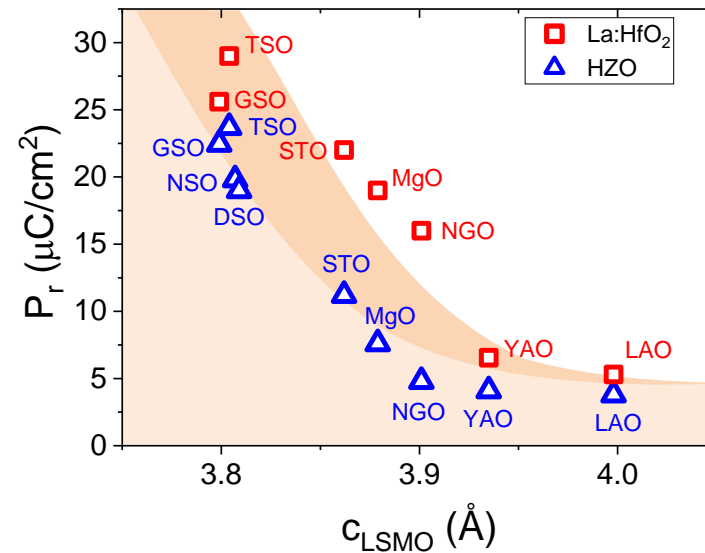
epi-La:HfO₂ →
LSMO →



Ferroelectric polarization

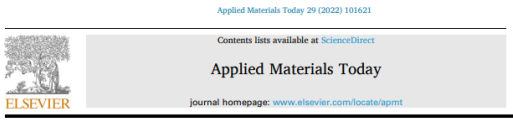


Zr doping produce similar effect



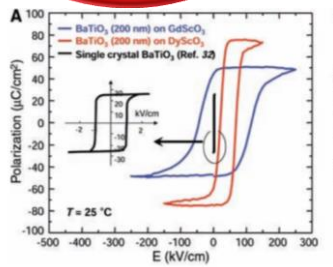
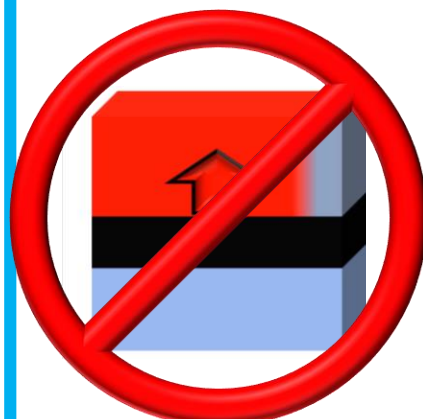
La doping → Greater Polarization

Results: strain/stress

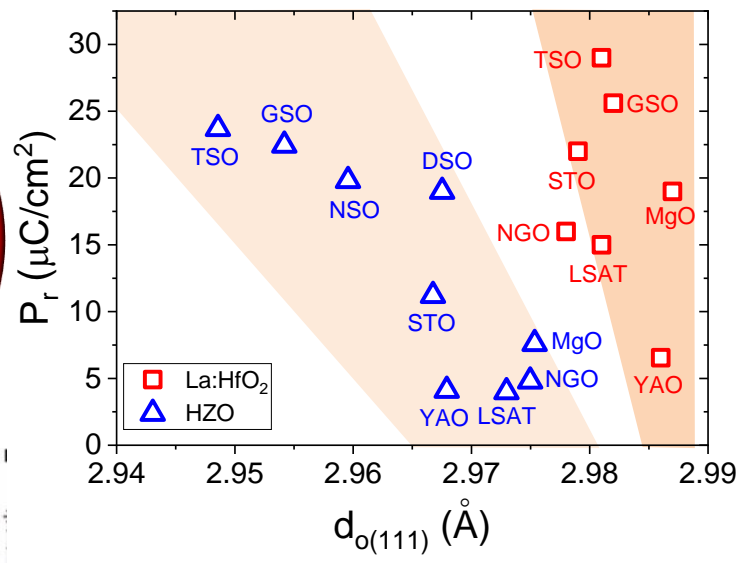


Synergetic contributions of chemical doping and epitaxial stress to polarization in ferroelectric HfO₂ films
 Tingfeng Song, Huan Tan, Anne-Claire Robert, Saül Estandia, Jaume Gàzquez, Florencio Sánchez, Ignasi Fina

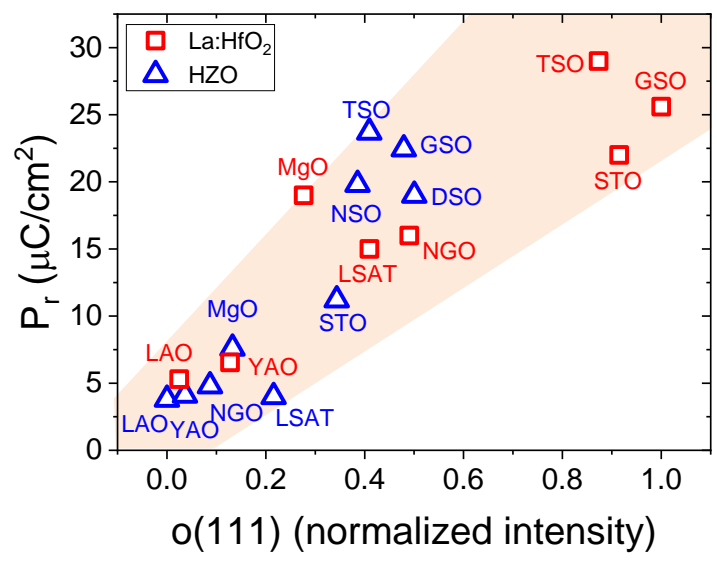
Song, ..., IF.
Applied Materials Today 29, 101621



Choi, et al., Science 306.5698 (2004): 1005-1009.



- No strain effects for different samples

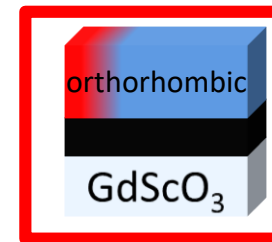
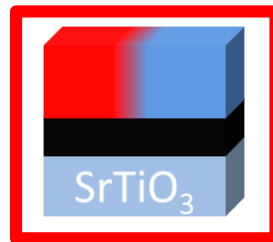
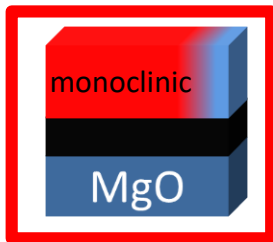


- Dependence on orthorhombic phase ratio amount prevails

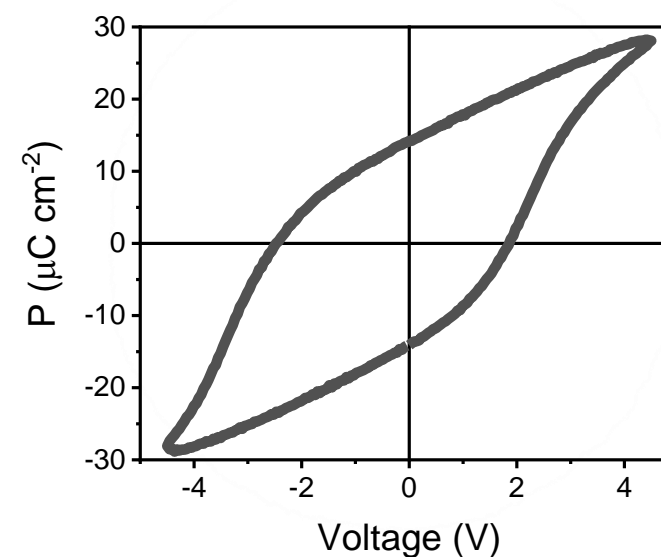
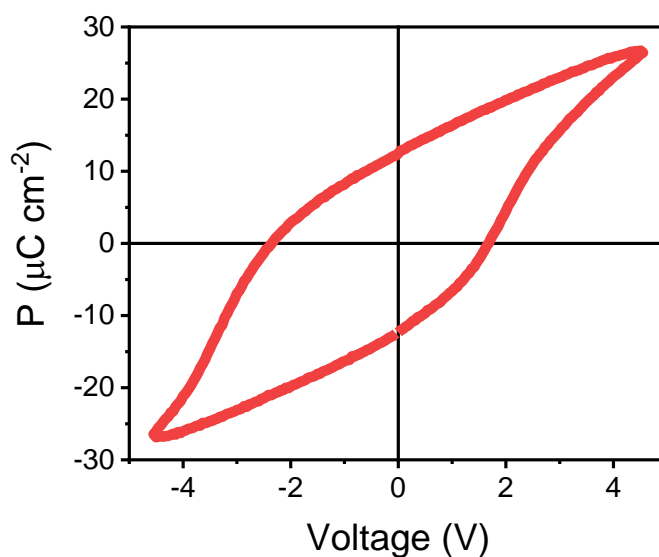
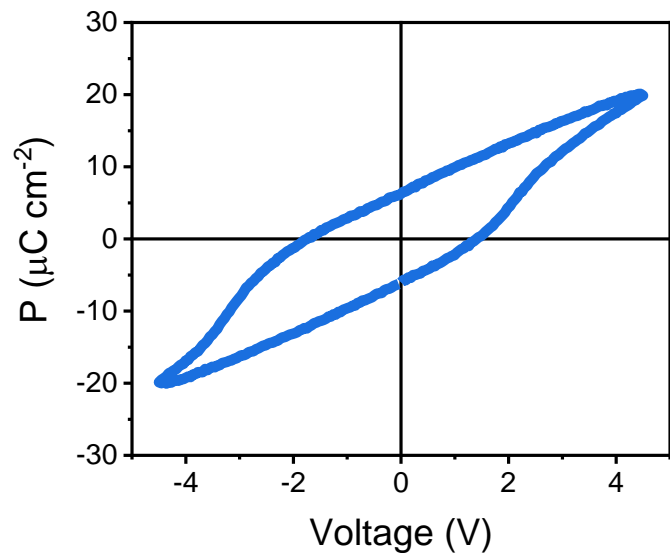


Results: polarization dependence on orthorhombic phase amount

epi-HZO →
LSMO →



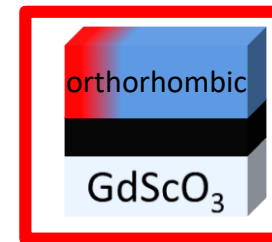
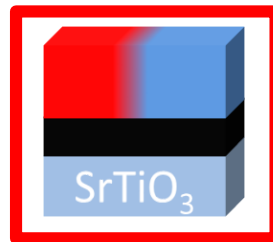
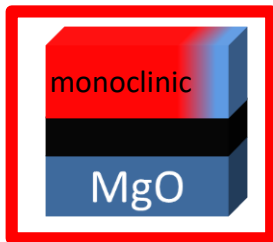
Orthorhombic phase amount



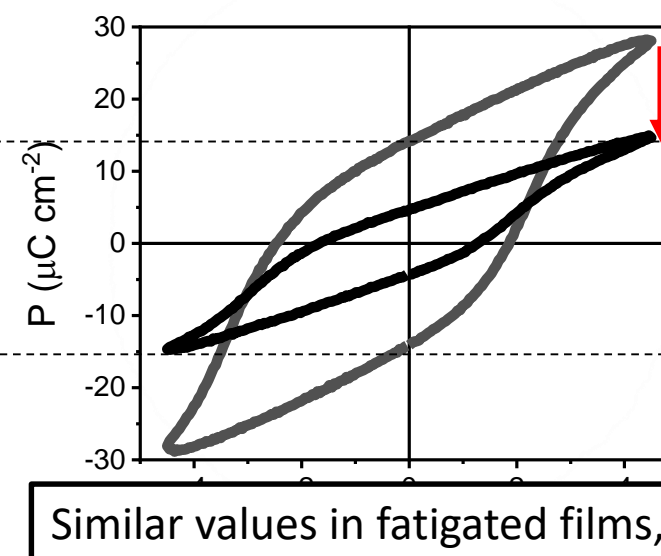
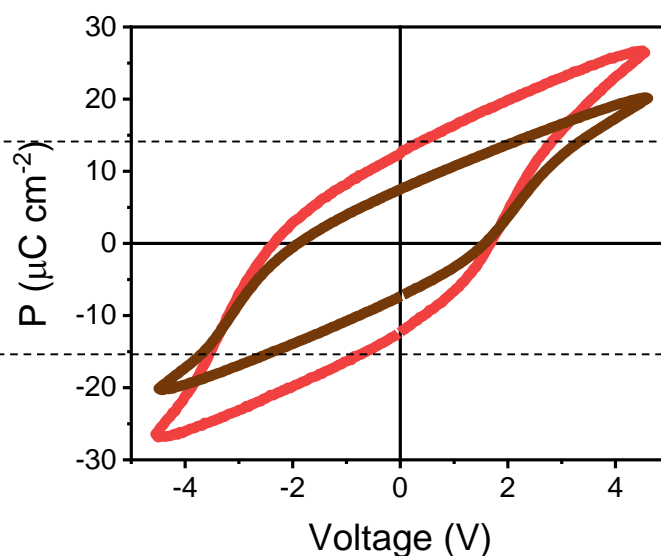
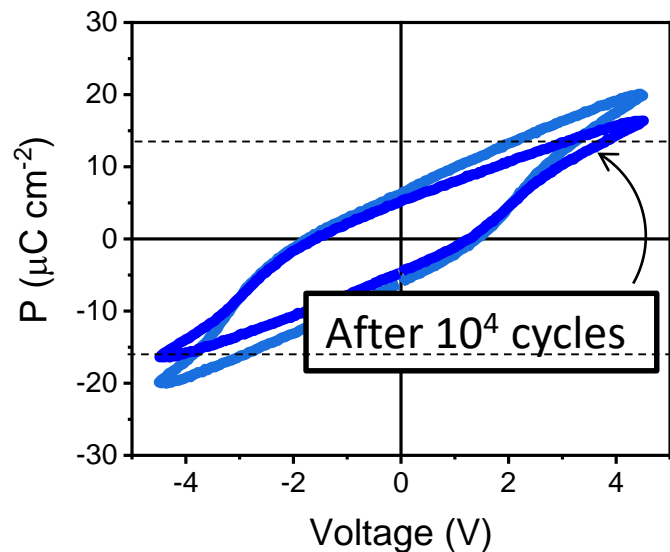
Ferroelectric polarization

Results: fatigue dependence on orthorhombic phase amount

epi-HZO →
LSMO →



Orthorhombic phase amount

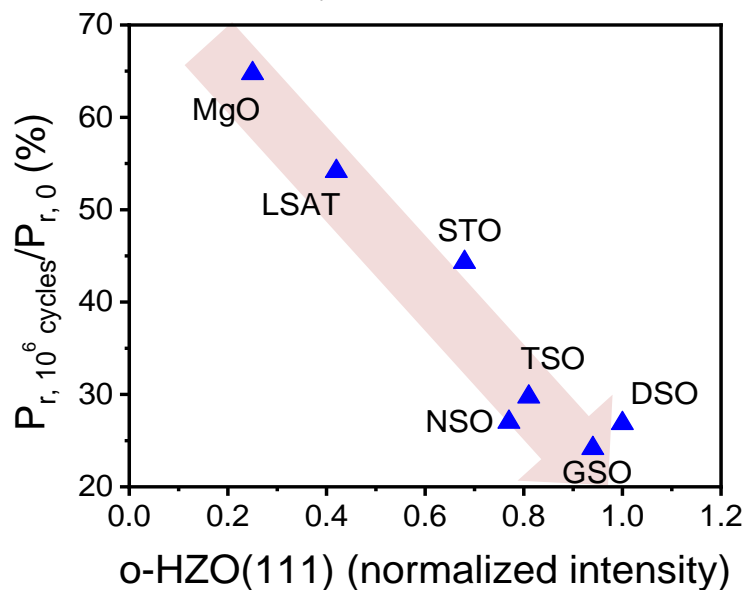
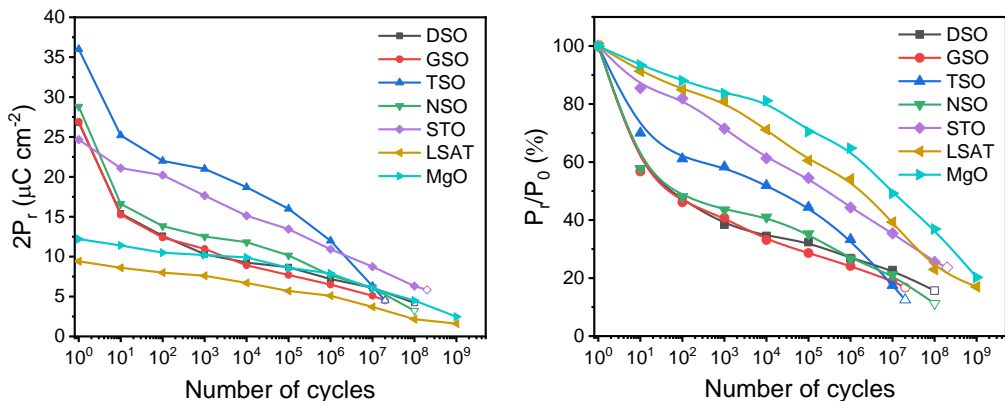


Ferroelectric polarization

Fatigue

Results: fatigue dependence on orthorhombic phase amount

Similar results are reproduced using a wide variety of substrates

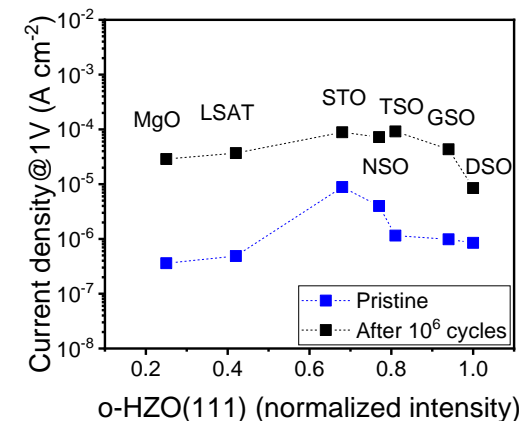


P_r and amount of orthorhombic phase

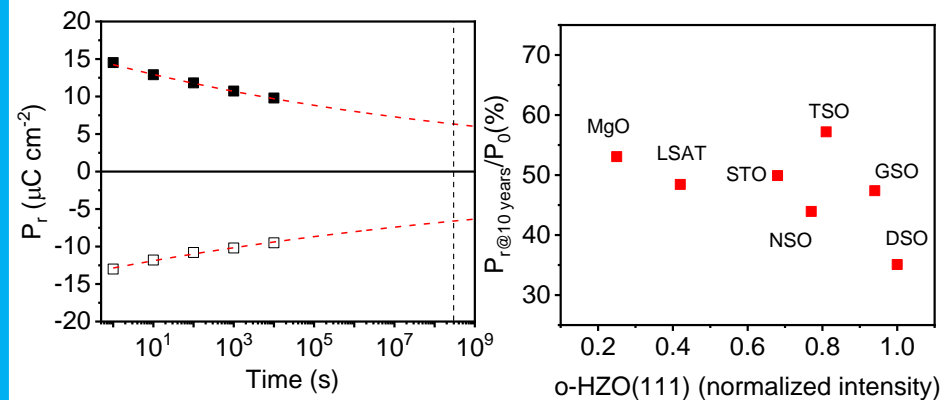
Endurance

Song, ..IF, FS.
Advanced Electronic Materials 8,
2100420 (2022)

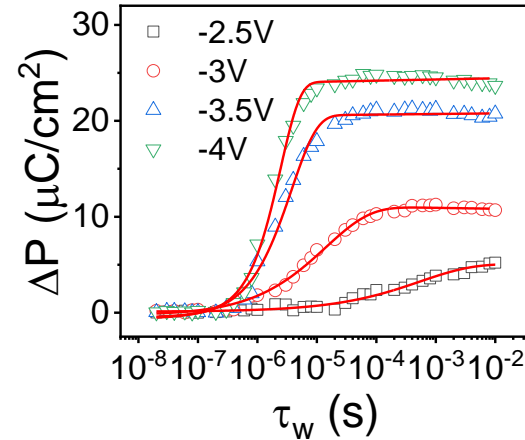
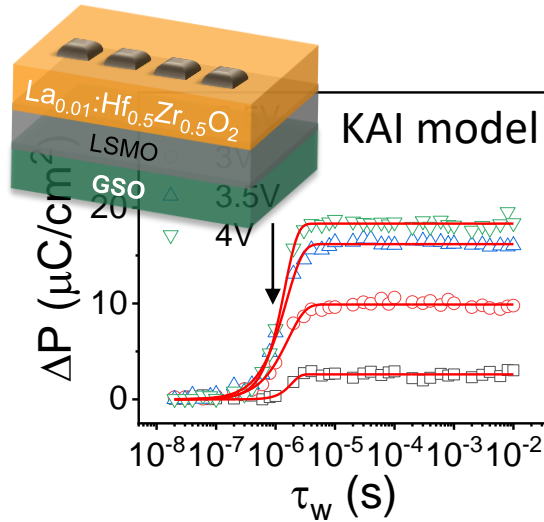
Important influence of leakage is disregarded



Retention is not importantly affected



Almost phase pure orthorhombic films

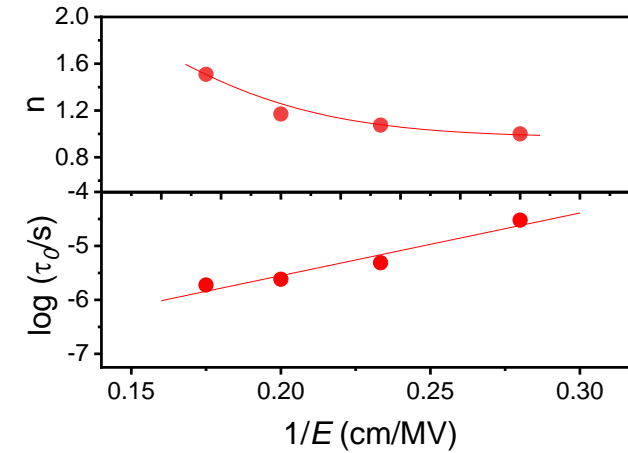


La:HZO/LSMO//GSO

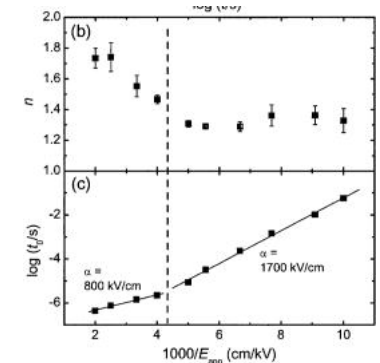
- For both polarities switching time is slow $1.4\mu\text{s}$
- For both polarities $n > 1$

$$P(t) / P_s = 1 - \exp \left[-(t / t_0)^n \right],$$

Dependence on electric field

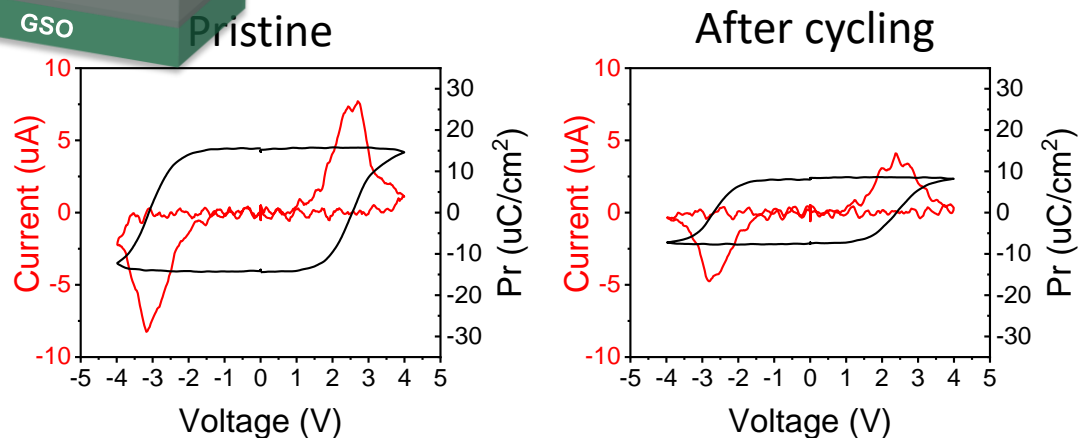
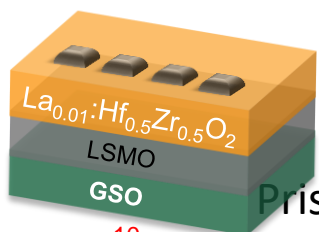


- As found in PZT switching time shortens with electric field and n decreases.
- KAI is valid for all fields



Y. W. So *et al.*, Appl. Phys. Lett. **86**, 092905 (2005).

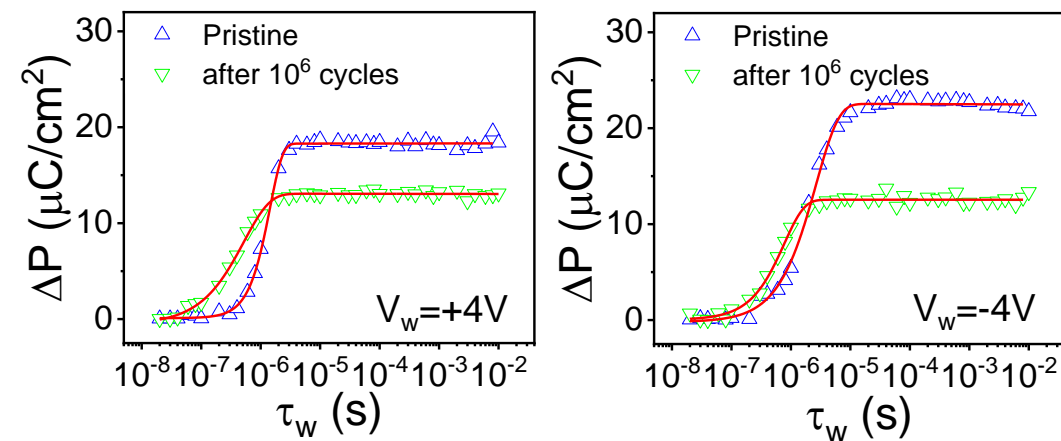
Fatigue effect on polarization



➤ After 10^6 cycles polarization decreases by $\approx 33\%$

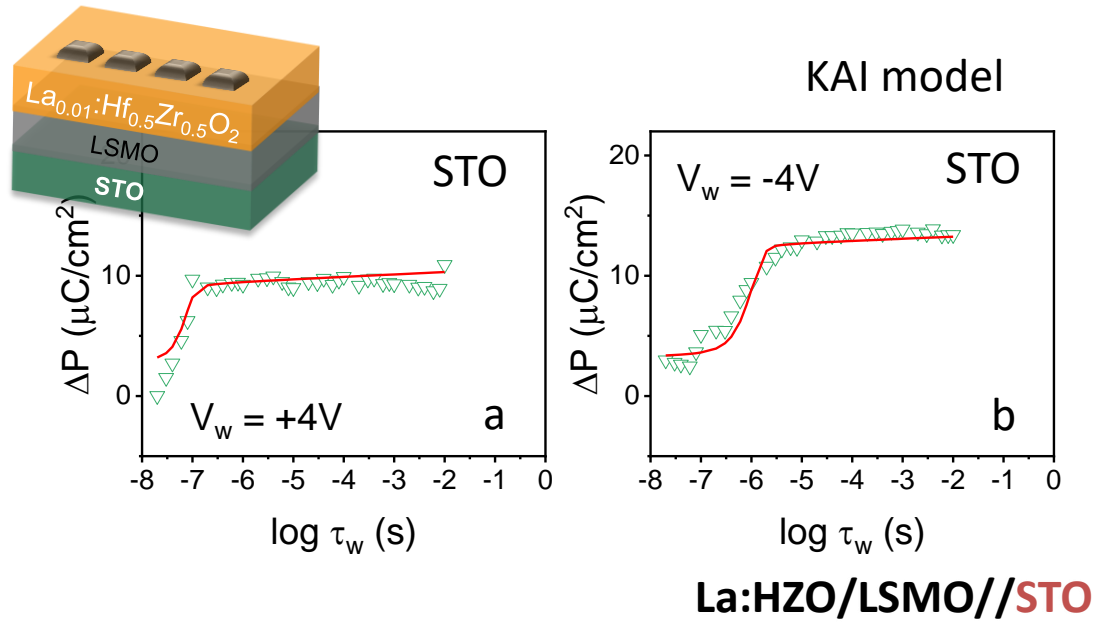
T. Song, et al., *Advanced Electronic Materials* 8 (1), 2100420 (2022)
M. Pešić et al., *Adv. Funct. Mater.* 26, 4601 (2016).
E. D. Grimley et al., *Adv. Electron. Mater.* 2, 1600173 (2016).

Fatigue effect on switching time



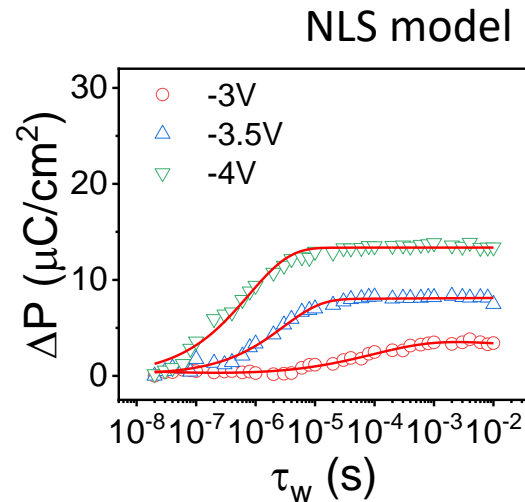
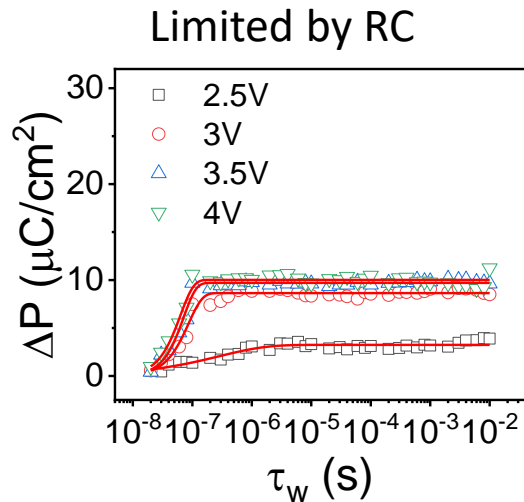
➤ After 10^6 cycles switching time decreases by $\approx 60\%$
➤ KAI is still valid

Mix phase pure orthorhombic films



- Fitting is not bad
- $n < 1$ = Unrealistic value

Mix phase pure orthorhombic films



La:HZO/LSMO//STO

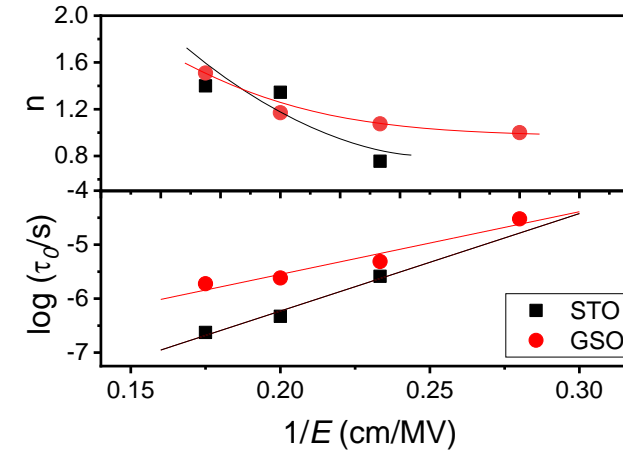
NLS model

$$\Delta P(t) = 2P_r \int_{-\infty}^{\infty} [1 - \exp\{-t/t_0\}^n] F(\log t_0) d(\log t_0)$$

$$F(\log t_0) = \frac{A}{\pi} \left[\frac{w}{(\log t_0 - \log t_1)^2 + w^2} \right]$$

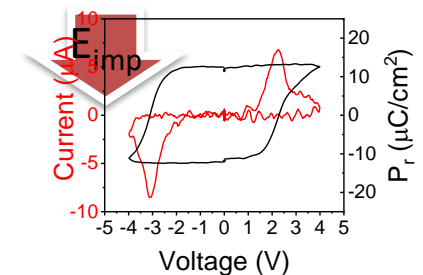
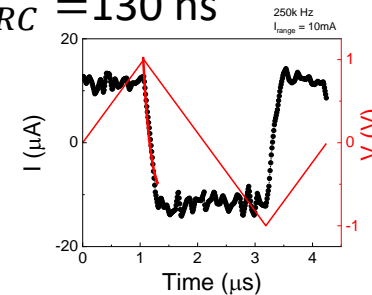
- Fitting with is not bad
- n values are >1
- For positive voltage switching dependence is limited by RC time

Dependence on electric field (for negative V)

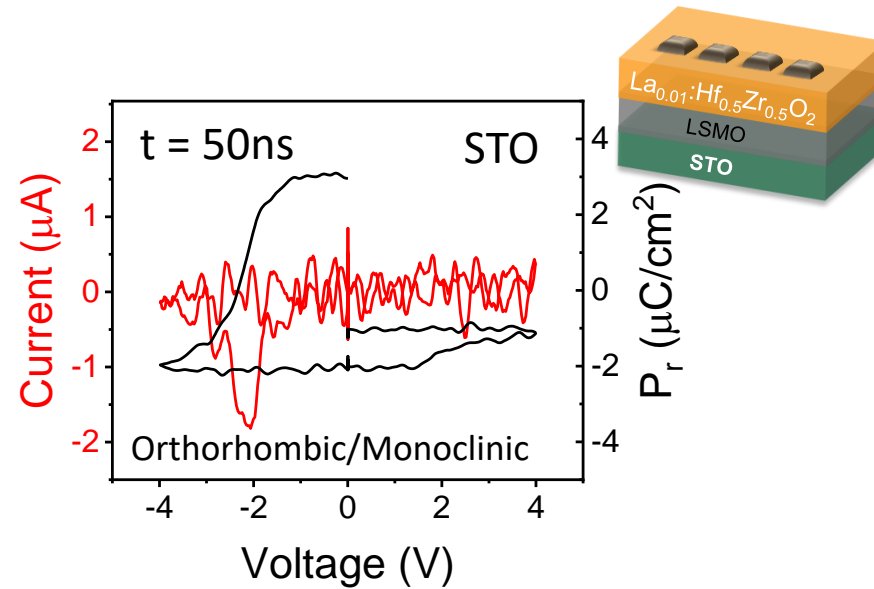
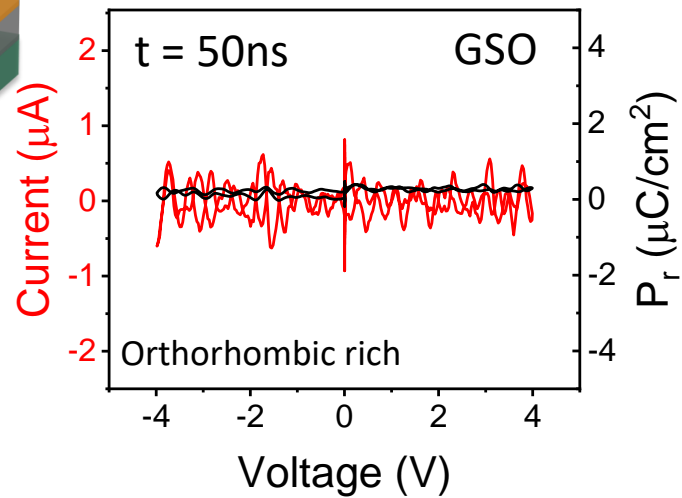
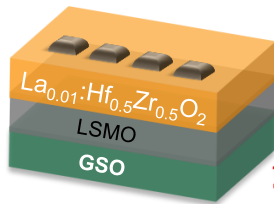


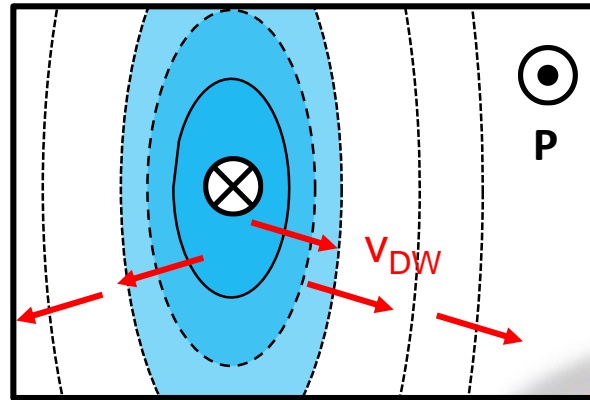
RC time constant (for positive V)

$\tau_{RC} = 130$ ns

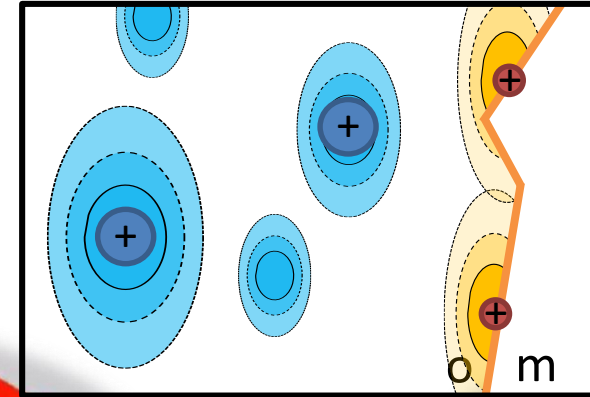


Mix phase film shows switched polarization using the same pulse width than the phase pure film

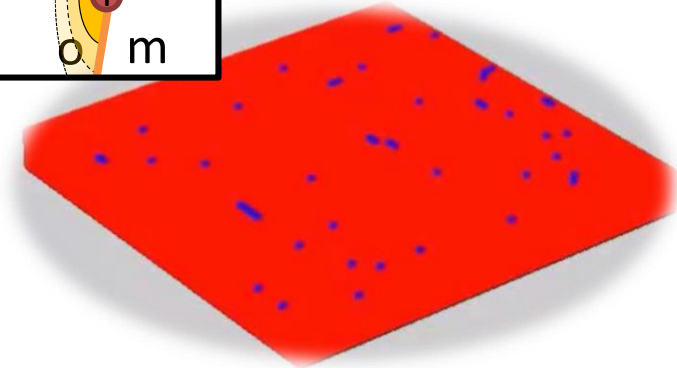
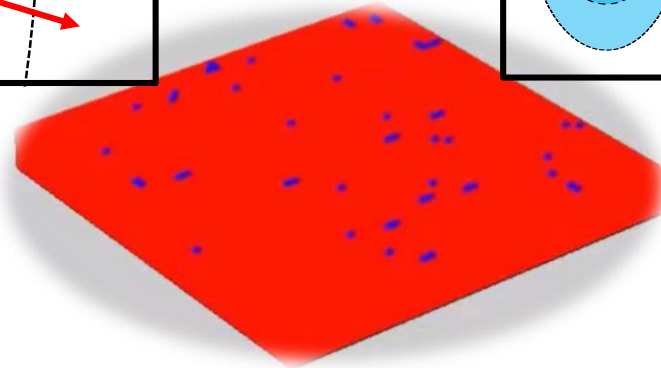




Phase pure



Mix phase



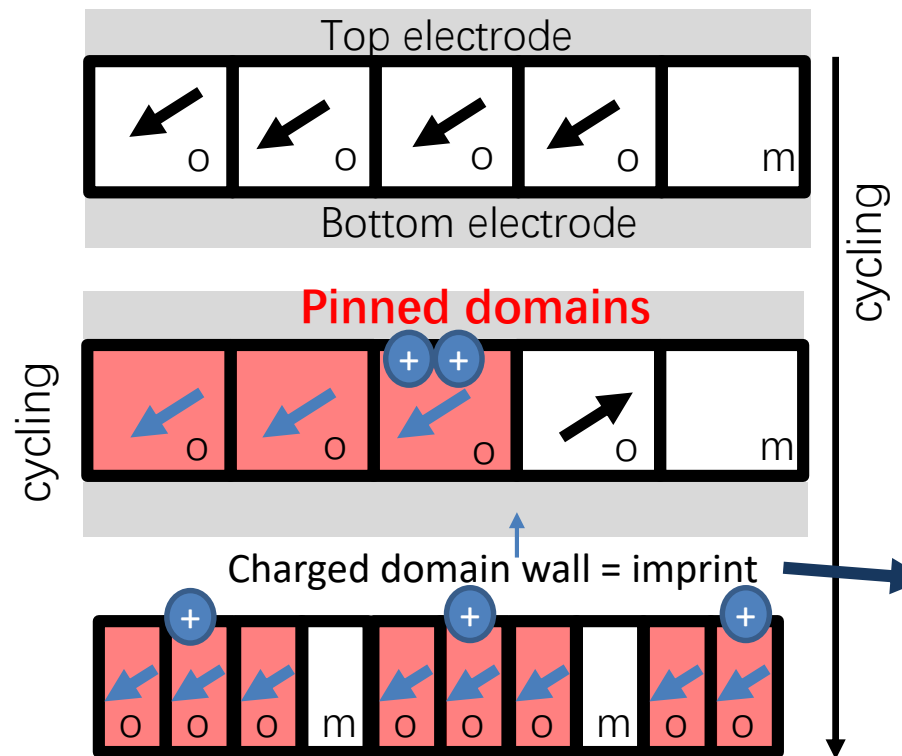
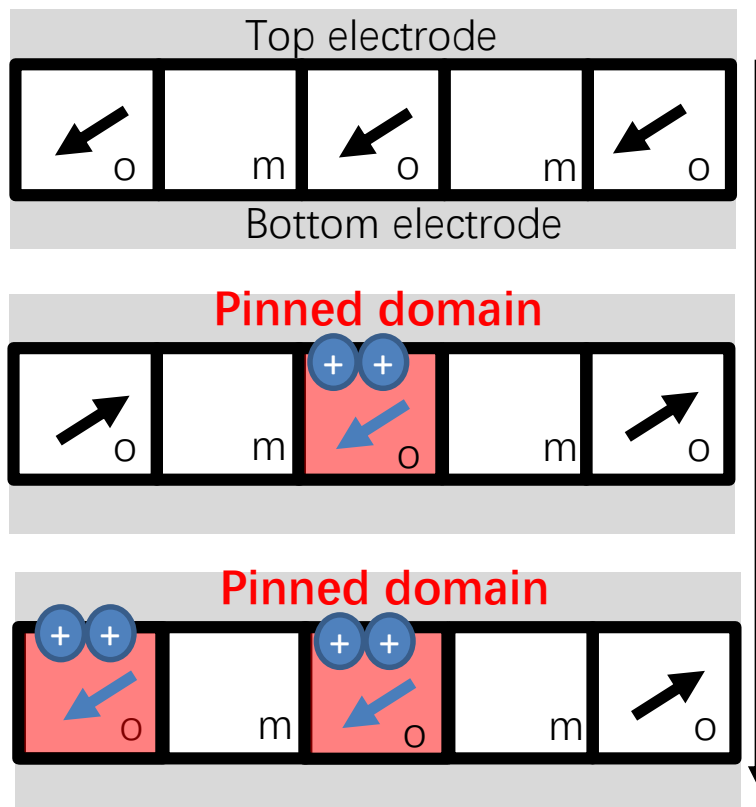
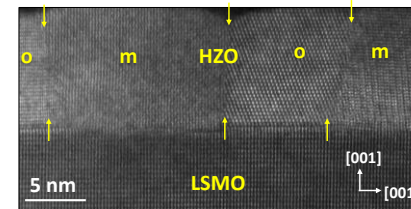
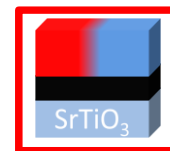
- The absence of defects/non-polar phase favors **2D domain** growth.
- In films on STO (coexisting orthorhombic/monoclinic phases), **charged defects/incoherent grain boundary** help on the initiation of the nucleation.

M. D. Glinchuk et al., J. Alloys Compd. 830, 153628 (2020).

P. Nukala et al., Science 372, 630 (2021).

Scenario: monoclinic grains are buffers for pinning propagation

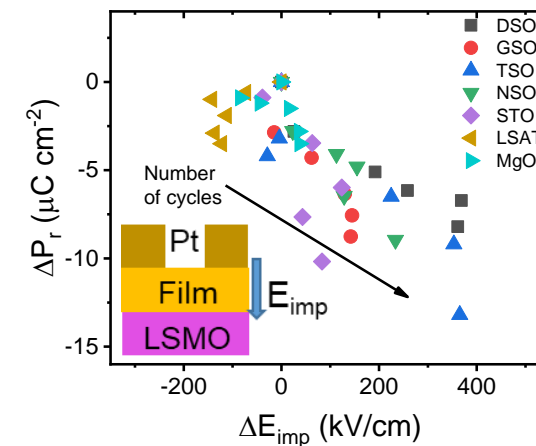
Grainy nature of epitaxial and polycrystalline films



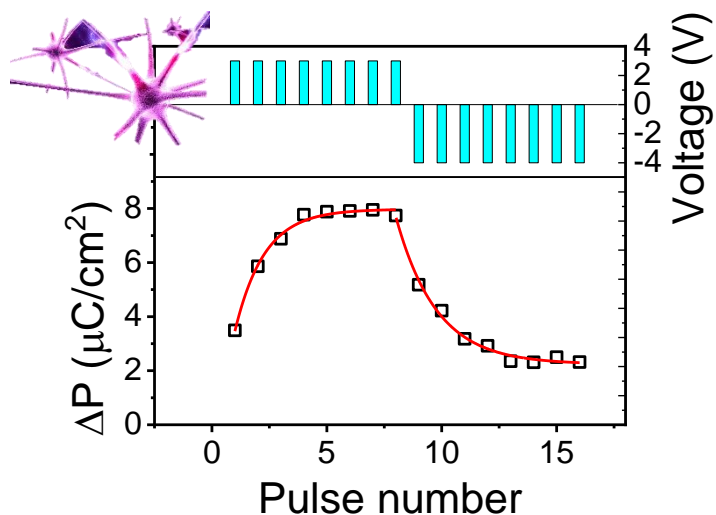
Assuming

- Oxygen vacancies
- At Pt/HZO interface

Imprint ant P reduction

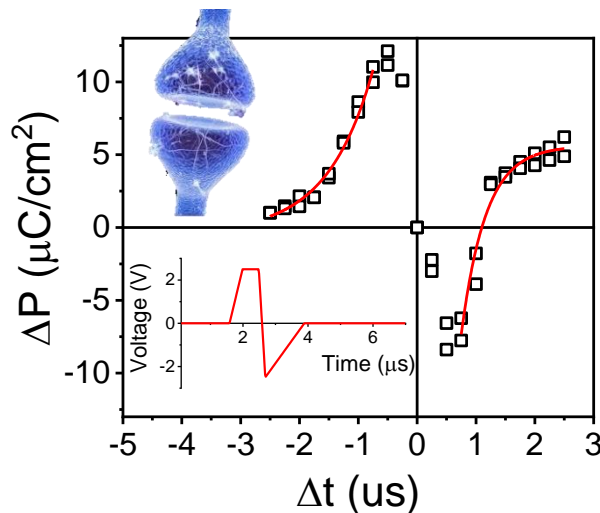


Beyond: neuromorphic-like behavior



50 ns pulse, +3V/-4V

Exponential coefficients of potentiation/depression, 1.2/1.8



750 ns

Shifted along ΔP (imprint)

Asymmetric Hebbian learning rule

$$\Delta\omega = Ae^{-\frac{\Delta t}{\tau}}$$

τ is the time constant of a STDP function

	$\Delta t > 0$	$\Delta t < 0$
[1]	6.53 μ s	9.28 μ s
[2]	35 μ s	22 μ s
This work	0.8 μ s	0.4 μ s

1.H. Y. Yoong et al., Adv. Funct. Mater. 28, 1806037 (2018).

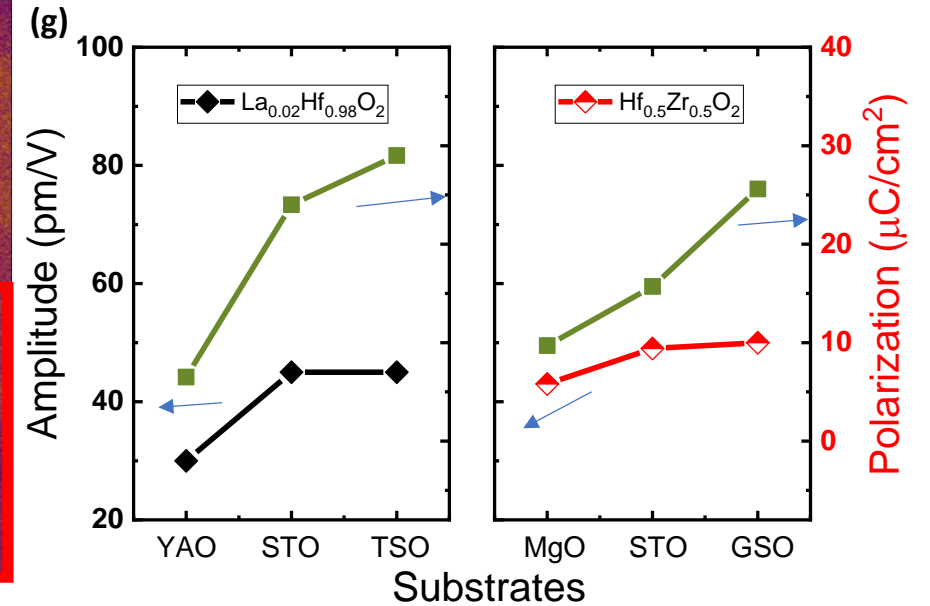
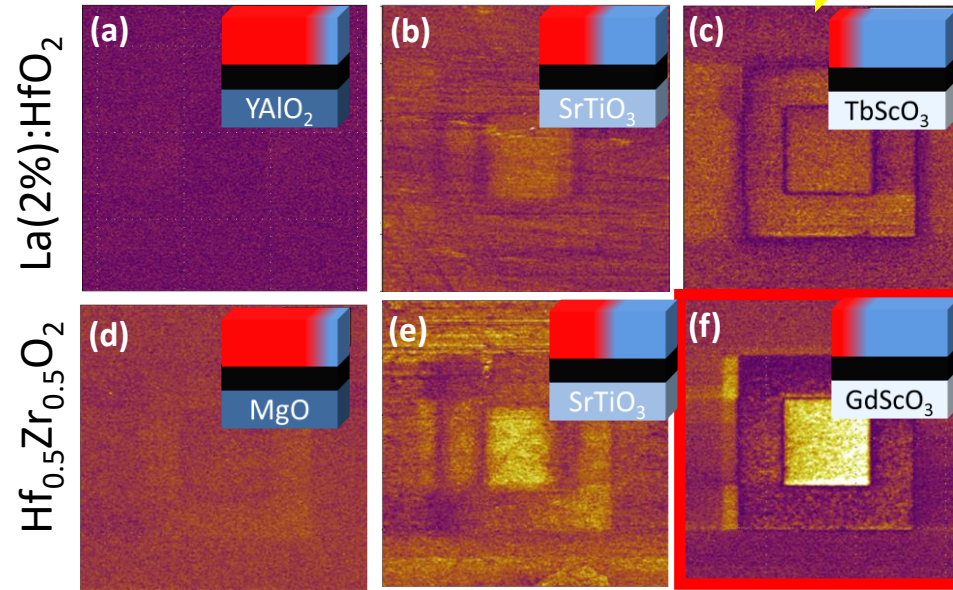
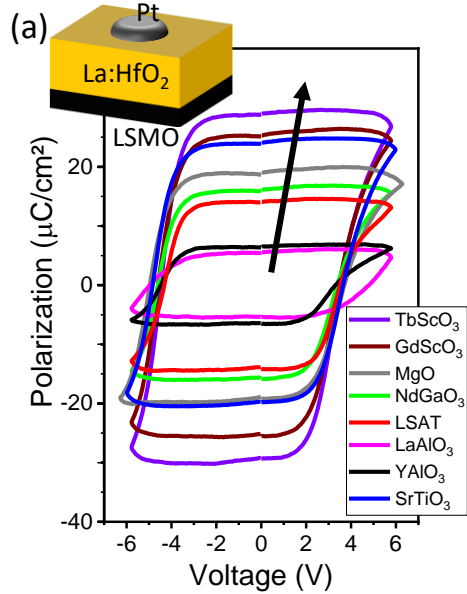
2.B. Max et al., ACS Appl. Electron. Mater. 2, 4023 (2020). 15U.

- Long term potentiation/depression and STPD characteristics at the shortest pulse time
- Both are non symmetric due to the defects

Beyond: Piezoelectric response

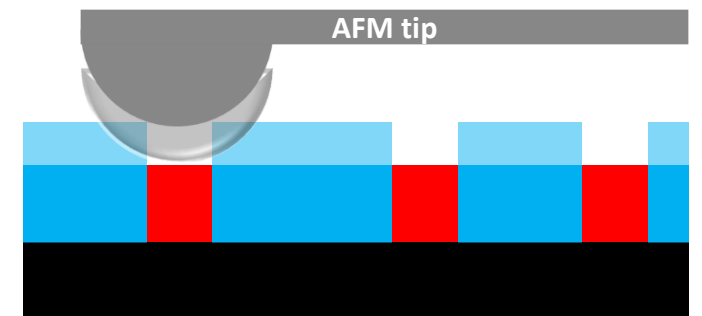
Amplitude piezo response

Orthorhombic phase amount



Brighter

- Orthorhombic phase shows larger piezoresponse
- Piezoresponse saturates due to tip size effect



Beyond: Piezoelectric response



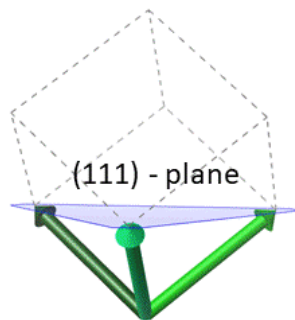
Cite this: DOI: 10.1039/d3tc01145c

Received 31st March 2023,
Accepted 8th May 2023

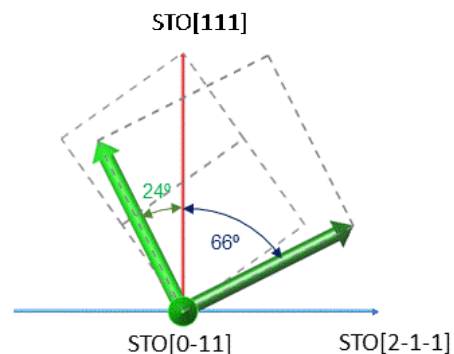
Vector piezoelectric response and ferroelectric domain formation in $\text{Hf}_{0.5}\text{Zr}_{0.5}\text{O}_2$ films†

Huan Tan, Tingfeng Song, Nico Dix, Florencio Sánchez* and Ignasi Fina*

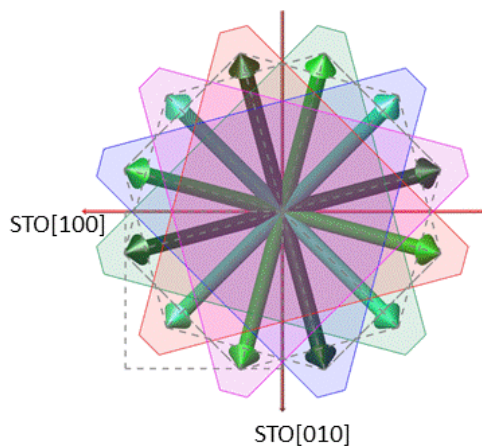
(a) HZO/LSMO/SrTiO₃(001) and (110)



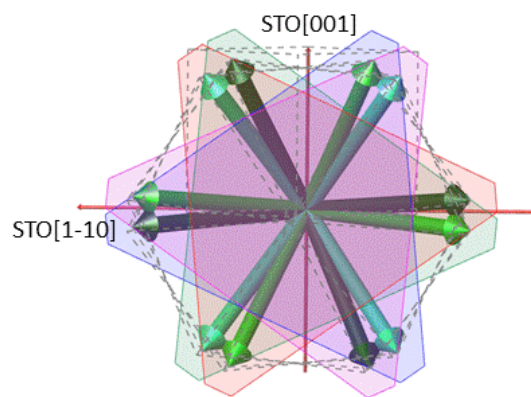
(b) HZO/LSMO/SrTiO₃(111)



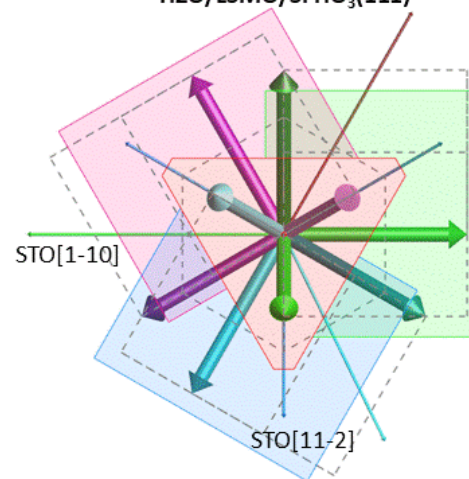
(c) HZO/LSMO/SrTiO₃(001)



(d) HZO/LSMO/SrTiO₃(110)

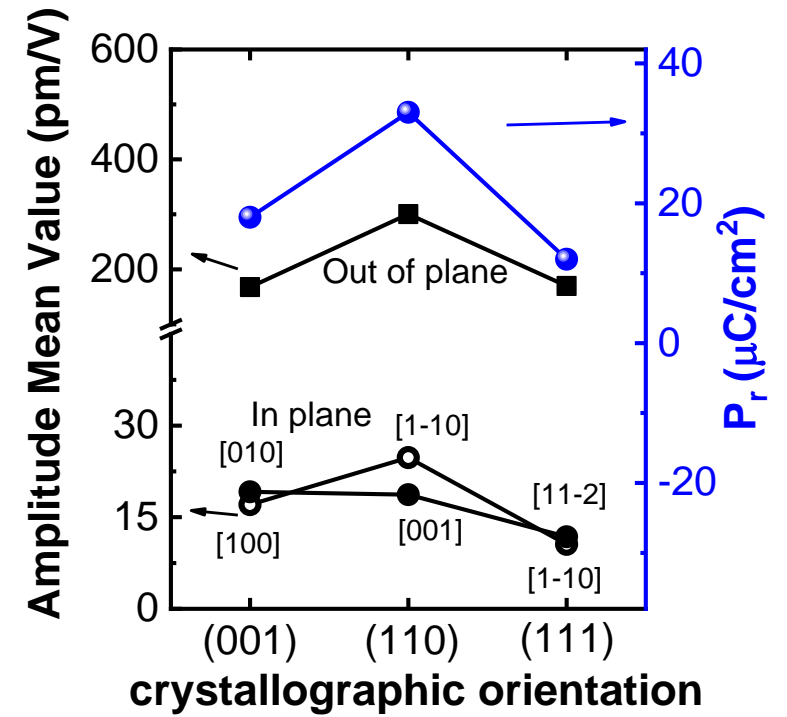
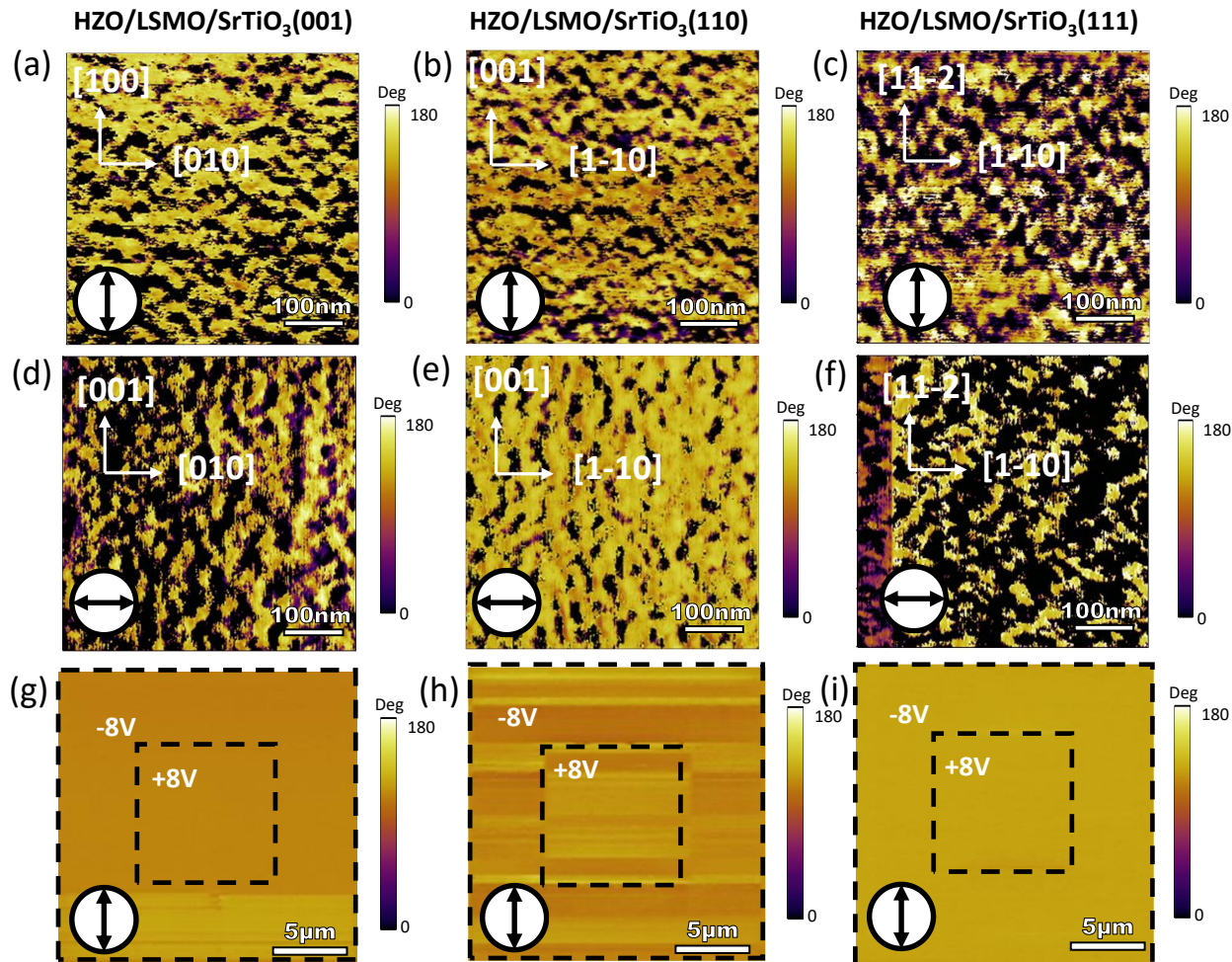


(e) HZO/LSMO/SrTiO₃(111)



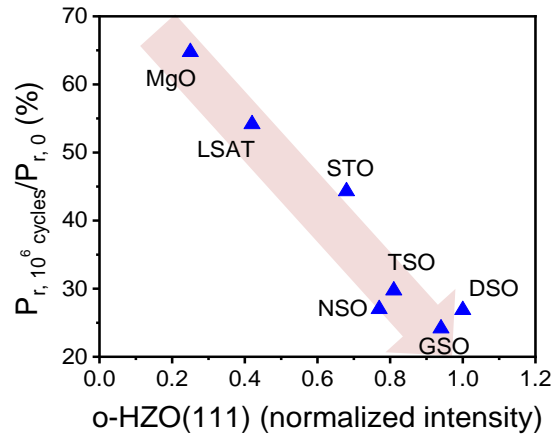
Tan, ...IF, 10.1039/D3TC01145C (Communication)
J. Mater. Chem. C, 2023, Advance Article
Song, ...IF,
Nanoscale 14, 2337 (2020)
Journal of Materials Chemistry C 10, 8407

Beyond: Piezoelectric response

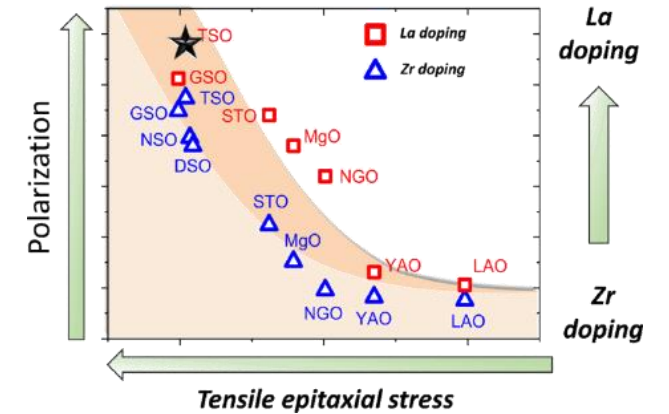


Conclusions

Ferroelectric polarization depends on orthorhombic phase amount (negligible strain effects)



Faster switching is observed in films with less orthorhombic phase



Endurance is not better in single-phase films

